

# Quantum Dot Laser

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**Johann Peter Reithmaier**

Technische Physik  
Institute of Nanostructure Technologies & Analytics (INA)  
University of Kassel

Tutorial for WWW.BRIGHTER.EU

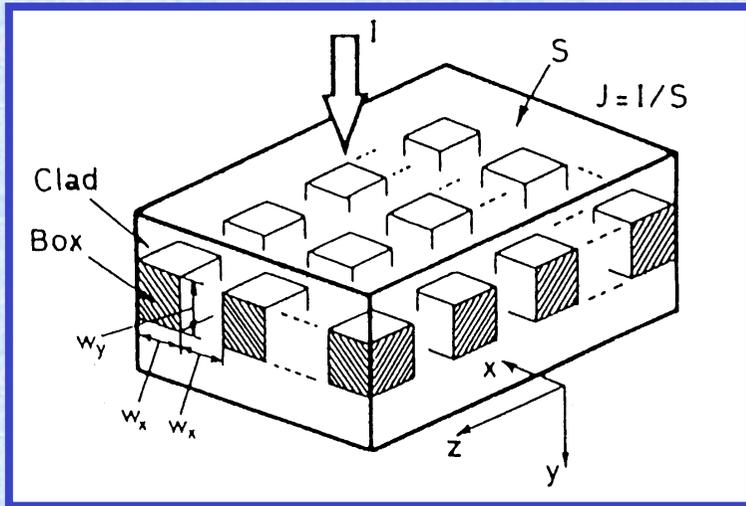
Lund, 29.6.2007



# Outline

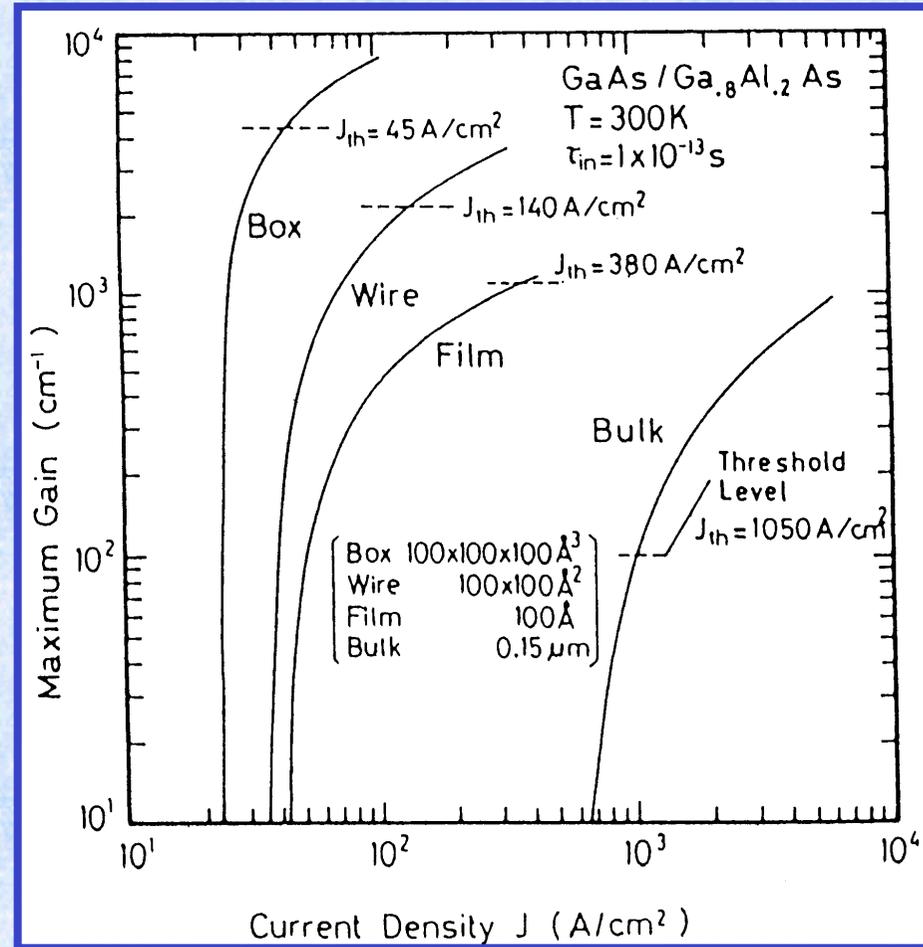
- Introduction and Some Basic Theory
  - density of states, gain function
  - dependence on dimensions
- Fabrication Technology
  - molecular beam epitaxy and Stranski-Krastanov growth mode
  - influence on geometry parameters
- Optical properties of real dots systems
  - single dot emission, higher order transitions
  - inhomogeneous linewidth, wetting layer
- Application examples of QD lasers
  - high power lasers (980 / 920 nm)
  - ultra-broadband lasers (1.55  $\mu\text{m}$ )
  - ultra-fast broadband SOAs (1.55  $\mu\text{m}$ )

# Quantum Dot Laser



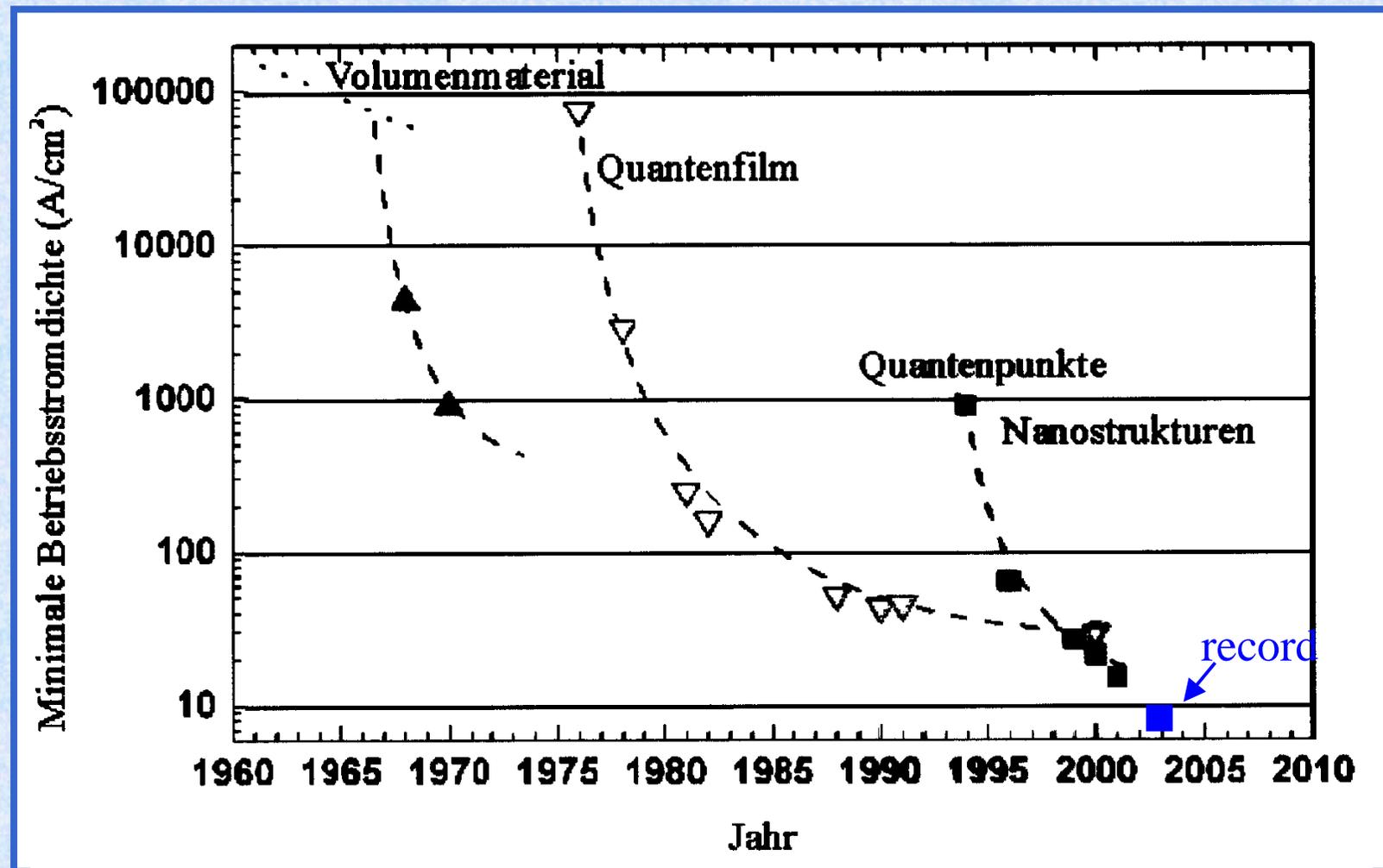
Due to higher density of states of quasi-zero dimensional systems a higher material and differential gain is expected in comparison to quantum well or bulk material

Also the transition matrix element has a higher value due to the improved overlapp of electron/hole wave functions.

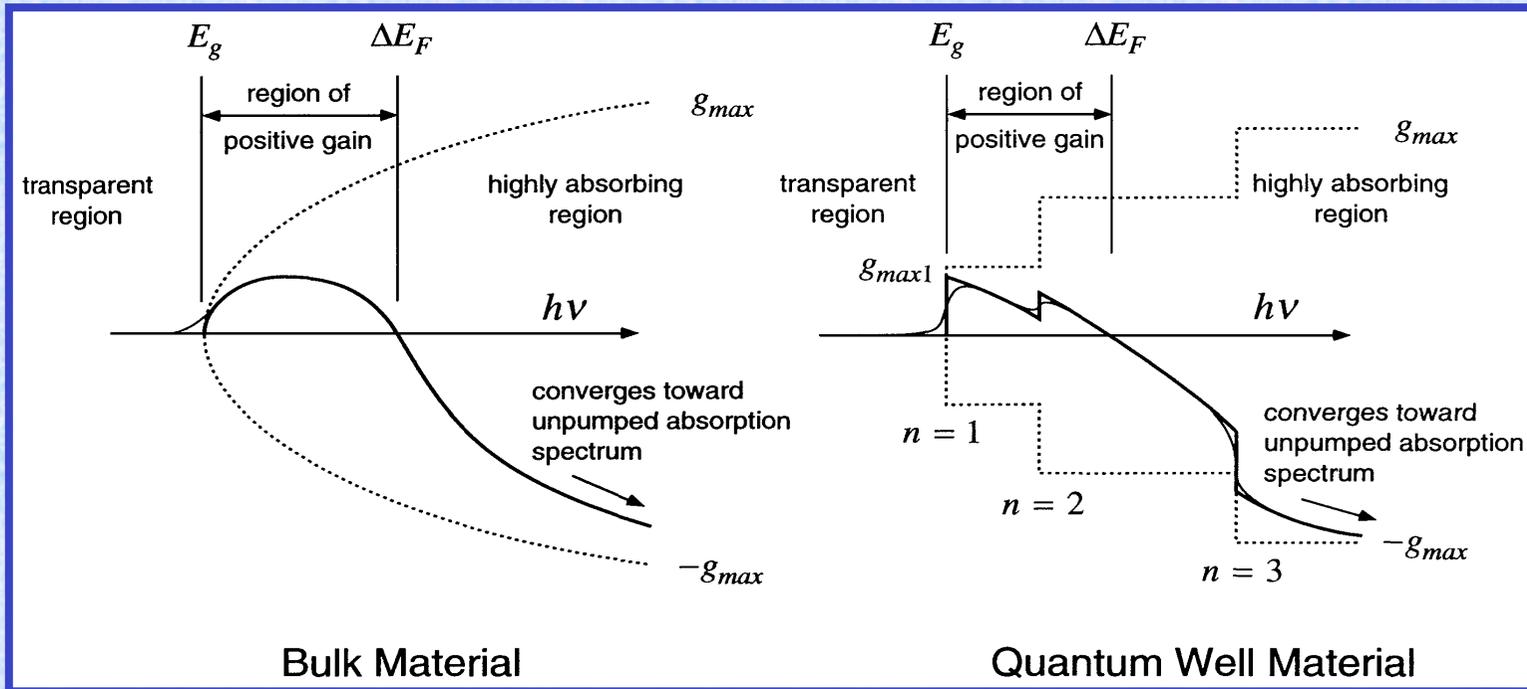


M. Asada et al., IEEE JQE 22, 1915 (1986)

# Time Evolution of Threshold Current Density



# Optical Material Gain



The gain can be changed by current injection between max. absorption (=  $-g_{max}$ ) and max. gain (=  $g_{max}$ ).

Spectral gain function

$$g(E_{21}) = \frac{\pi q^2 \hbar}{n \epsilon_0 c m_0^2} \frac{1}{\hbar \omega_{21}} \underbrace{|M_T(E_{21})|^2}_{\text{transition matrix element}} \underbrace{\rho_r(E_{21})}_{\text{reduced density of state function}} (f_2 - f_1) = g_{max}(E_{21}) \cdot (f_2 - f_1)$$

L.A. Coldren, S.W. Corzine, "Diode Lasers and Photonic Integrated Circuits", Wiley, 1995.

$\rho_r(E_{21})$  reduced density of state function  
 $M_T(E_{21})$  transition matrix element

# Density of State Function for 2D Case

Number of states within circle of radius  $p$ :

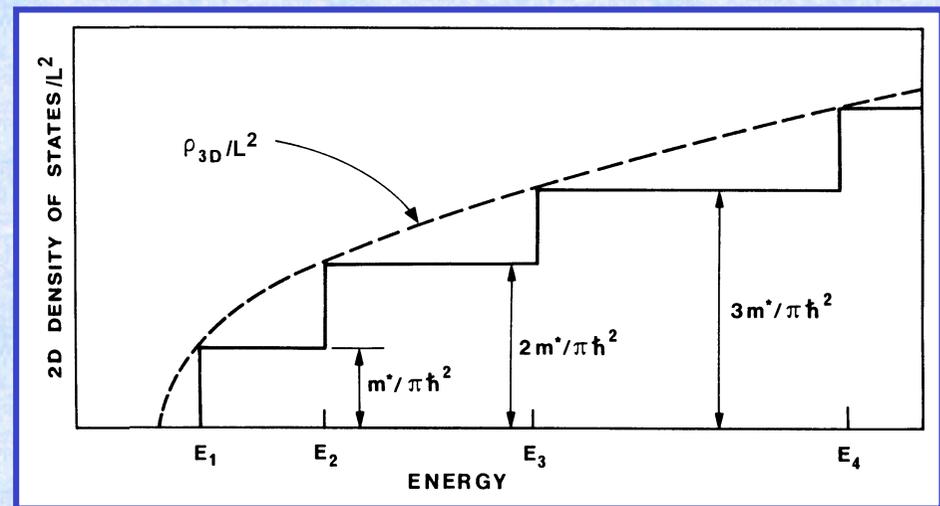
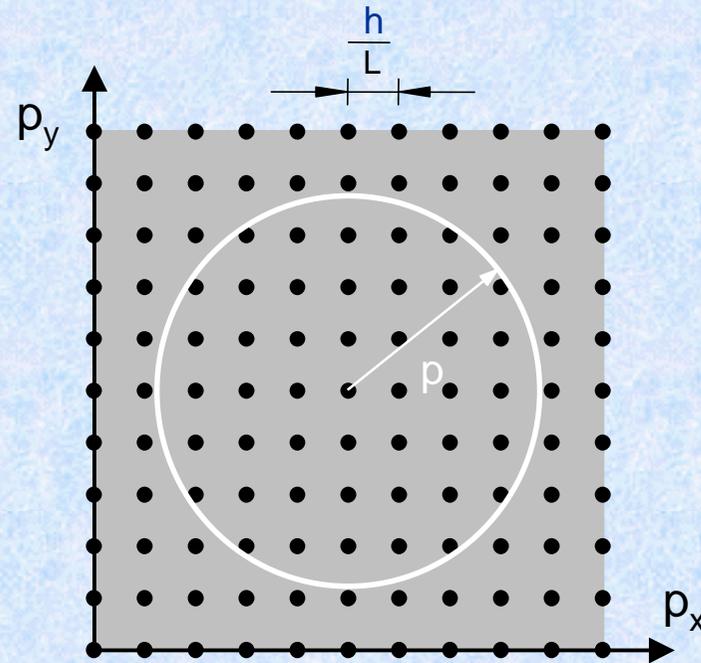
$$N(p) = 2 \times \frac{\pi p^2}{(h/L)^2}$$

→ Density of state function in momentum space:

$$D(p) = \frac{dN}{dp} = \frac{4\pi p A}{h^2}$$

Density of state function as function of energy:

$$D(E) = \sum_{i=1}^n \frac{m^* L^2}{\pi \hbar^2} \Theta(E - E_i)$$



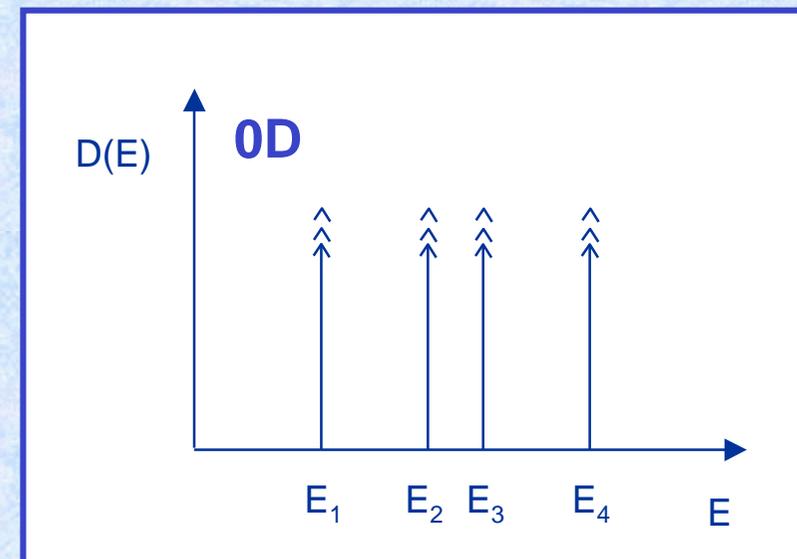
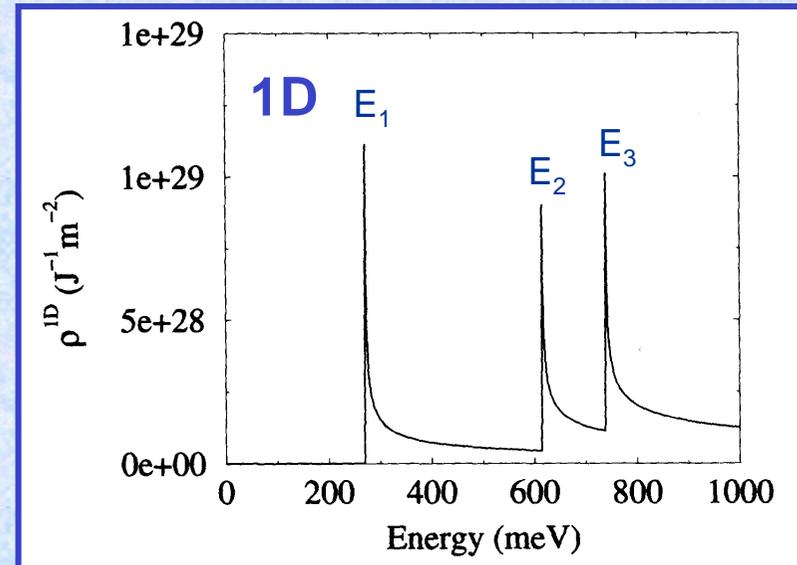
# Density of State Functions in 1D and 0D

$$D(E) = \sum_{i=1}^n \sqrt{\frac{2m^*L^2}{\hbar^2}} \frac{1}{\pi \cdot \sqrt{E - E_i}} \Theta(E - E_i)$$

with Heaviside function  $\Theta(E)$ ,  
L as length of structure and  
 $m^*$  as effective mass of particle

$$D(E) = \sum_{i=1}^n \alpha_i \cdot \delta(E - E_i)$$

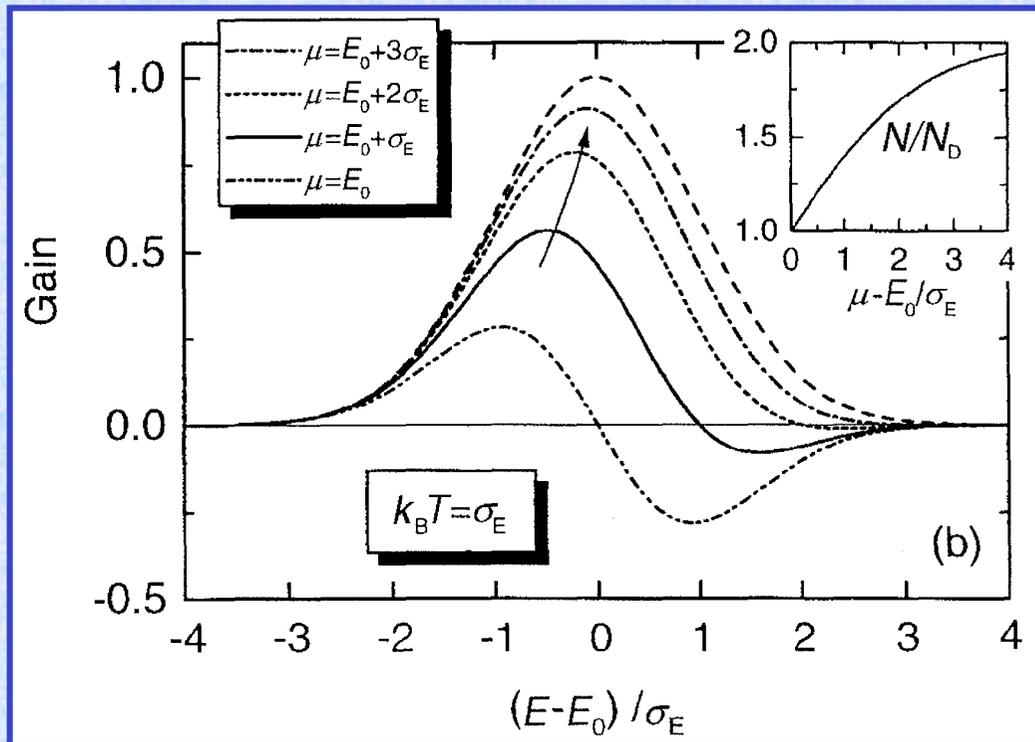
with  $\alpha_i$  as degeneracy  
(= 2 for s orbitals)



# Density of State Dependence on Dimensionality

dimension	density of state $D(p)$	density of state $D(E)$
3D	$\frac{8\pi V}{h^3} \cdot (p)^2$	$\frac{V}{2\pi^2} \left( \frac{2m^*}{\hbar^2} \right)^{3/2} \cdot (E)^{1/2}$
2D	$\frac{4\pi A}{h^2} \cdot (p)^1$	$\frac{A}{2\pi} \left( \frac{2m^*}{\hbar^2} \right)^1 \cdot (E)^0$
1D	$\frac{2L}{h} \cdot (p)^0$	$\frac{L}{\pi} \left( \frac{2m^*}{\hbar^2} \right)^{1/2} \cdot (E)^{-1/2}$
0D	$\delta(p - p_i)$	$\delta(E - E_i)$

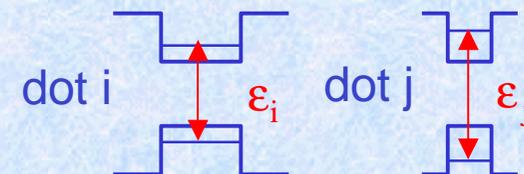
# Spectral Gain Function of QD Material



Spectral gain for inhomogeneously broadened dot ensemble with one confined state

$\mu$  = chemical potential

$\sigma_E$  = inhomogeneous linewidth



$$g(\hbar\omega) = \frac{\pi q^2 \hbar}{n \epsilon_0 c m_0^2 \hbar \omega} \int |M_T(\hbar\omega)|^2 \frac{2}{V_0} P(\epsilon, \sigma_E) [f_c(\epsilon, E_{Fc}) - f_v(\epsilon, E_{Fv})] \underbrace{\frac{\Gamma_{in} / \pi}{(\hbar\omega - \epsilon)^2 + \Gamma_{in}^2}}_{\text{Lorentzian function of homogeneous linewidth}} d\epsilon$$

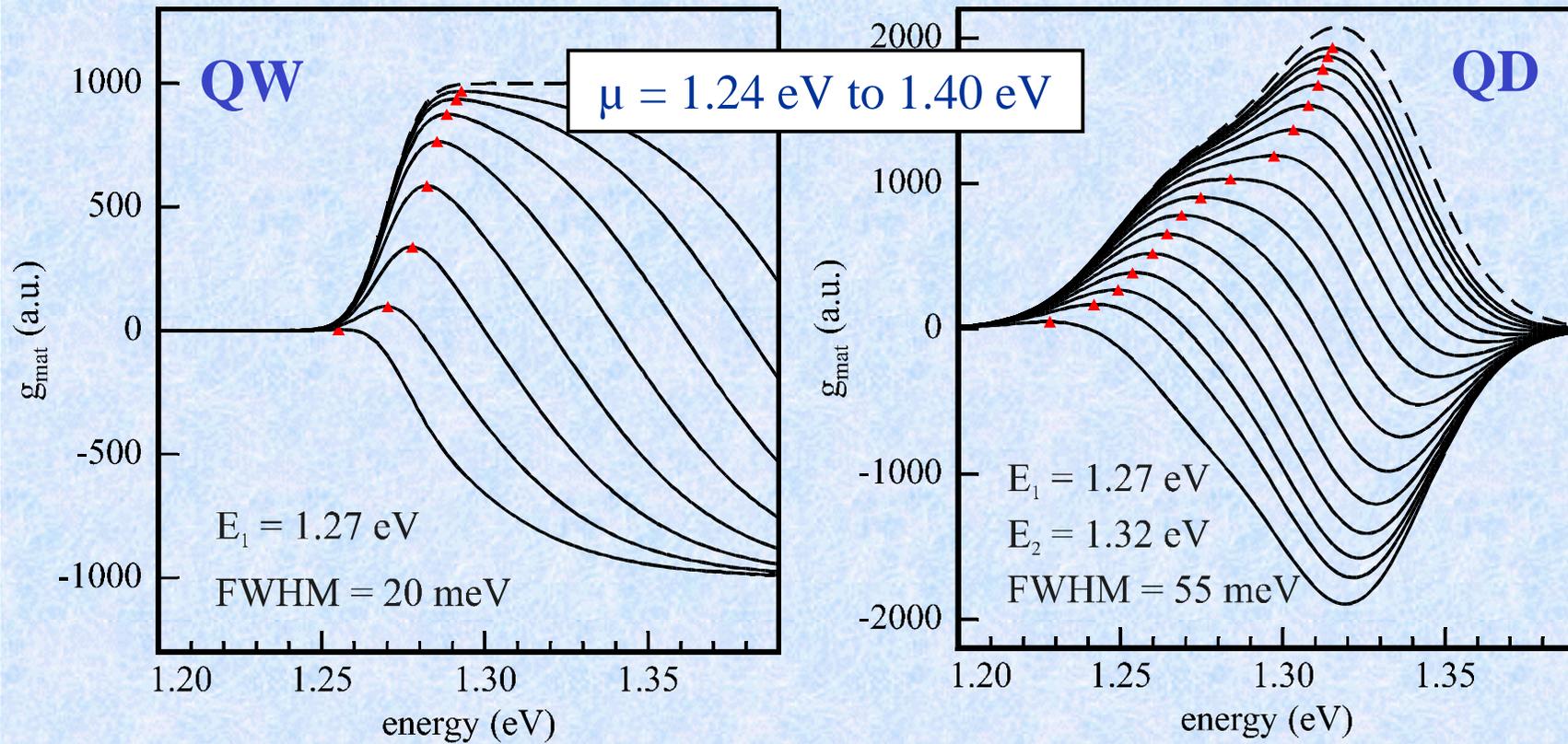
with inhomogeneous ensemble linewidth:

$$P(\epsilon, \sigma_E) = \frac{1}{\sigma_E \sqrt{2\pi}} \exp\left[-\frac{(\epsilon - E_g - E_0)^2}{2\sigma_E^2}\right]$$

Lorentzian function of homogeneous linewidth

# Gain Engineering with QD Properties

$$g_{mat}(E) \propto D(E) \times f(E, \mu)$$



- Reduced blue shift due to high total gain

- After saturation of first transition large blue shift

F. Klopf et al., Photonics West, 2002

# Comparison with Experimental Data

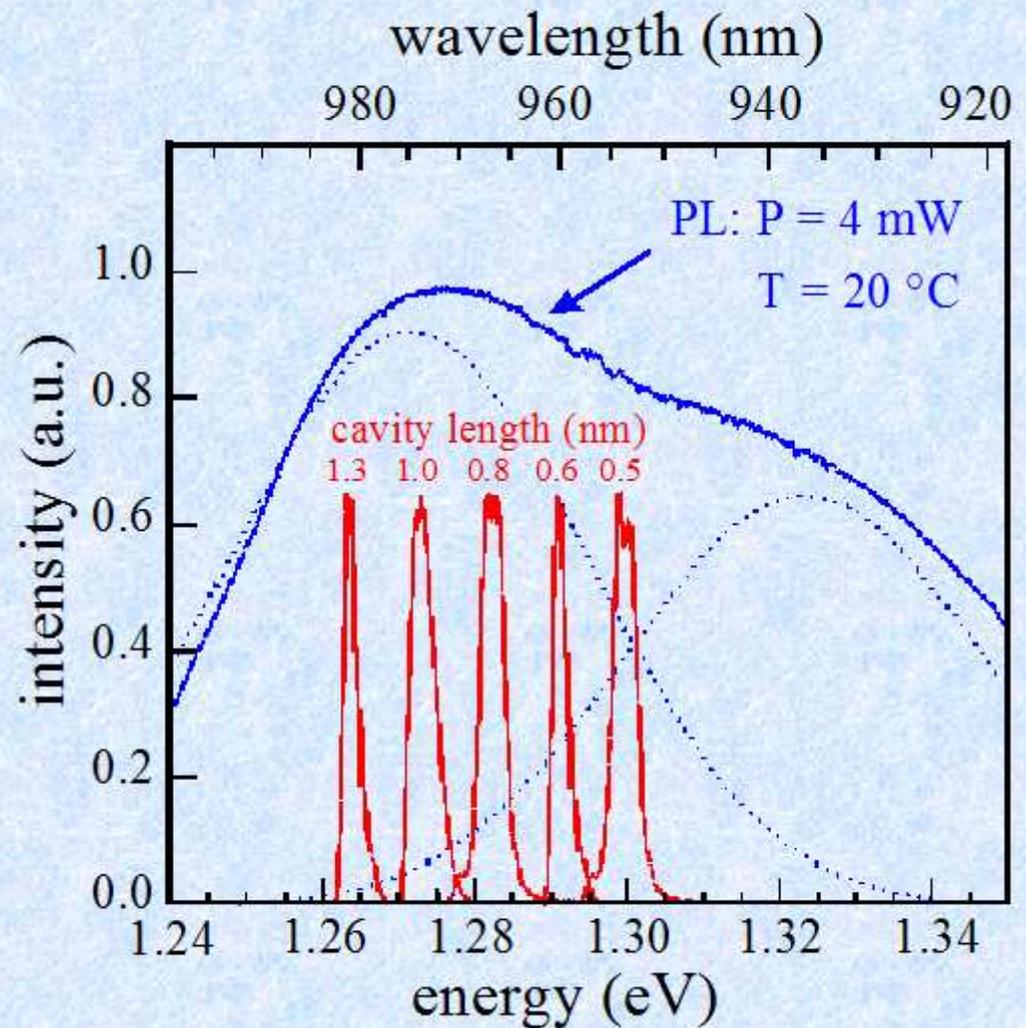
Gain profile evaluated by high power excitation PL experiment

→ Spectral gain function

Control of mirror losses by variation of cavity length

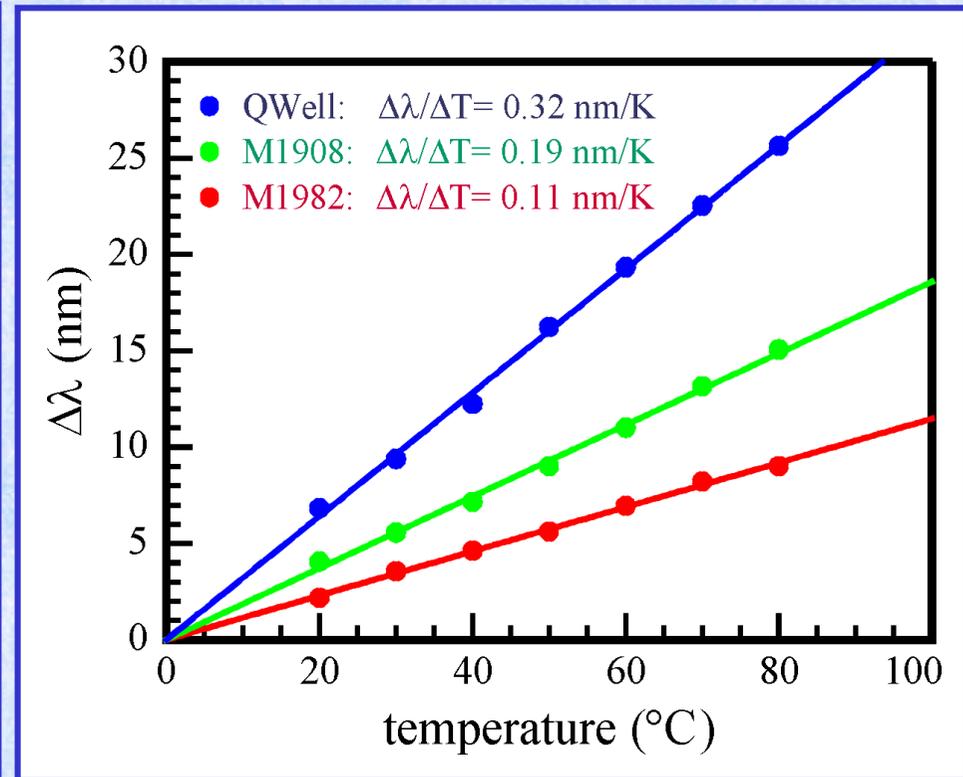
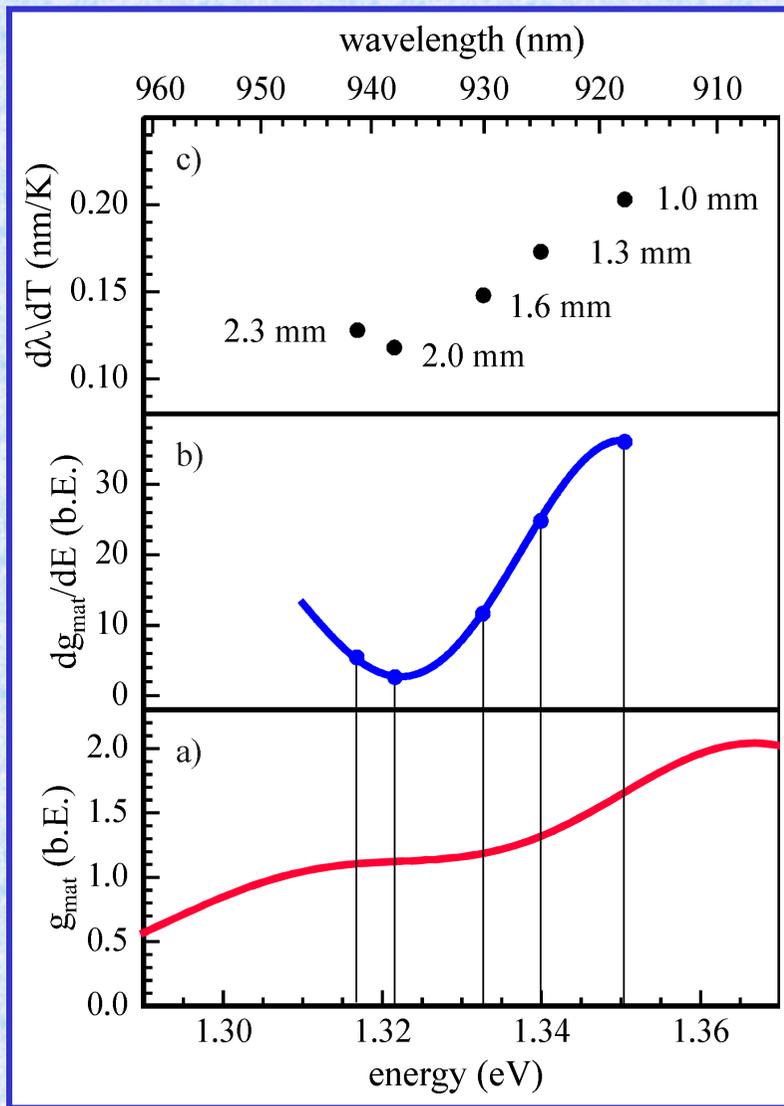
→ Variation of emission wavelength

$$g_{th} = \alpha_i + \frac{1}{L} \ln \left( \frac{1}{\sqrt{R_1 R_2}} \right)$$



F. Klopf et al., Photonics West, 2002

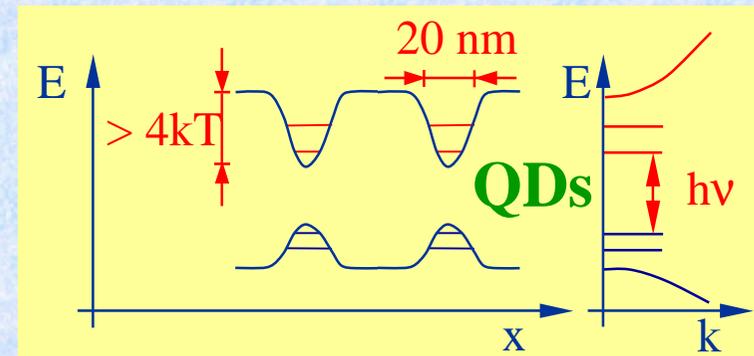
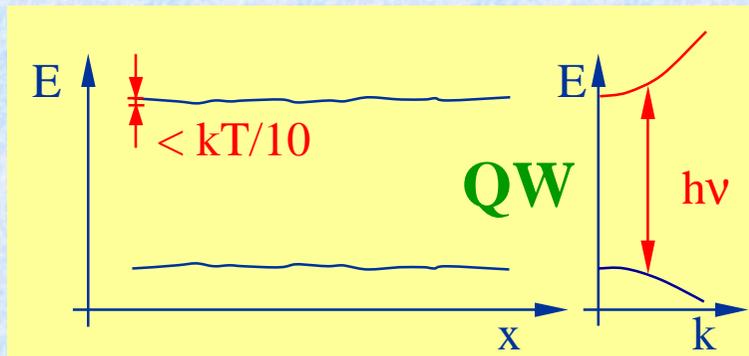
# Temperature Stability



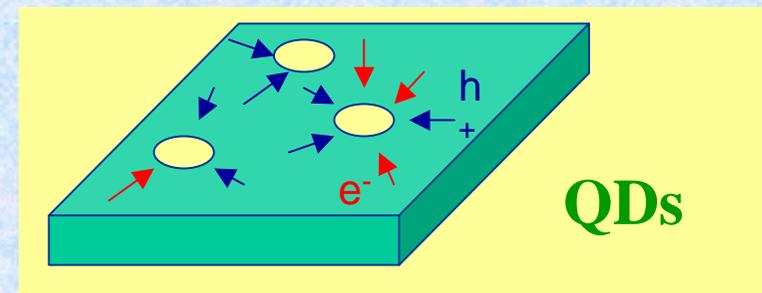
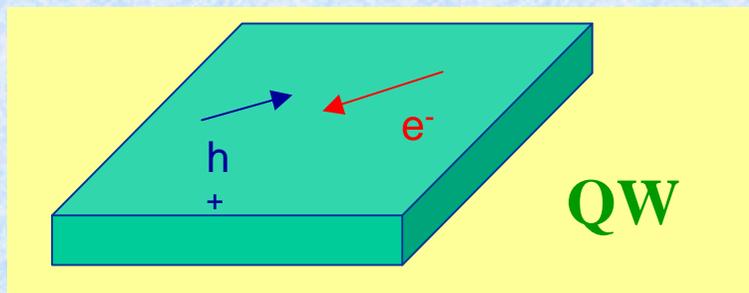
- Flat gain profile
- Very low temperature dependence of emission wavelength of 0.11 nm/K (best value for BA lasers: 0.09 nm/K)

Basic effect described in  
F. Klopf et al., APL 81, 217 (2002)

# Specific Properties of QD Gain Material



Discrete energy levels: → high density of states, no temperature dependence



reduced diffusion:

→ no diffusion to surfaces

reduced active volume:

→ low absorption, low inversion densities

refractive index decoupled  
from carrier density

→ low chirp

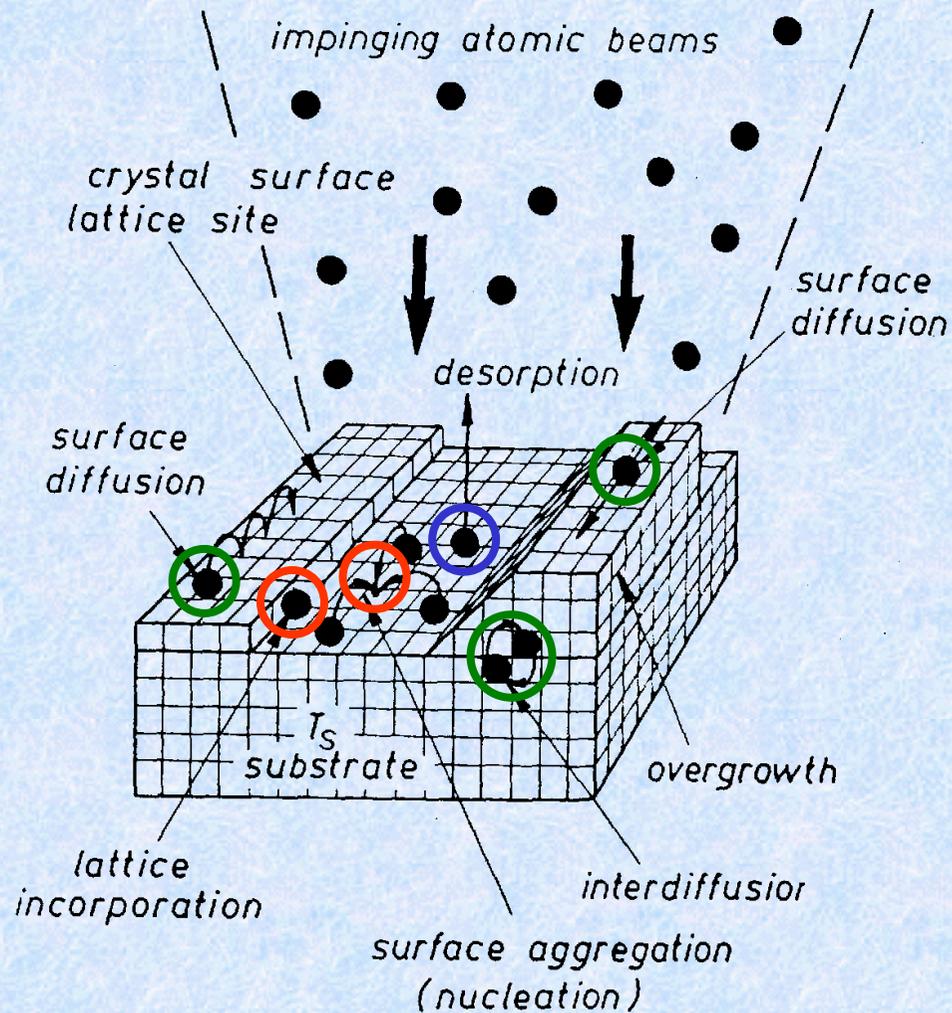
+ symmetric gain function:

$$\alpha = - \frac{4\pi}{\lambda} \frac{dn/dN}{dg/dN}$$

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# Basic Phenomena During Epitaxial Growth

## Example: Molecular Beam Epitaxy



### Positive Growth

- Incorporation into crystal steps
- Spontaneous nucleation

### Negative Growth

- Desorption

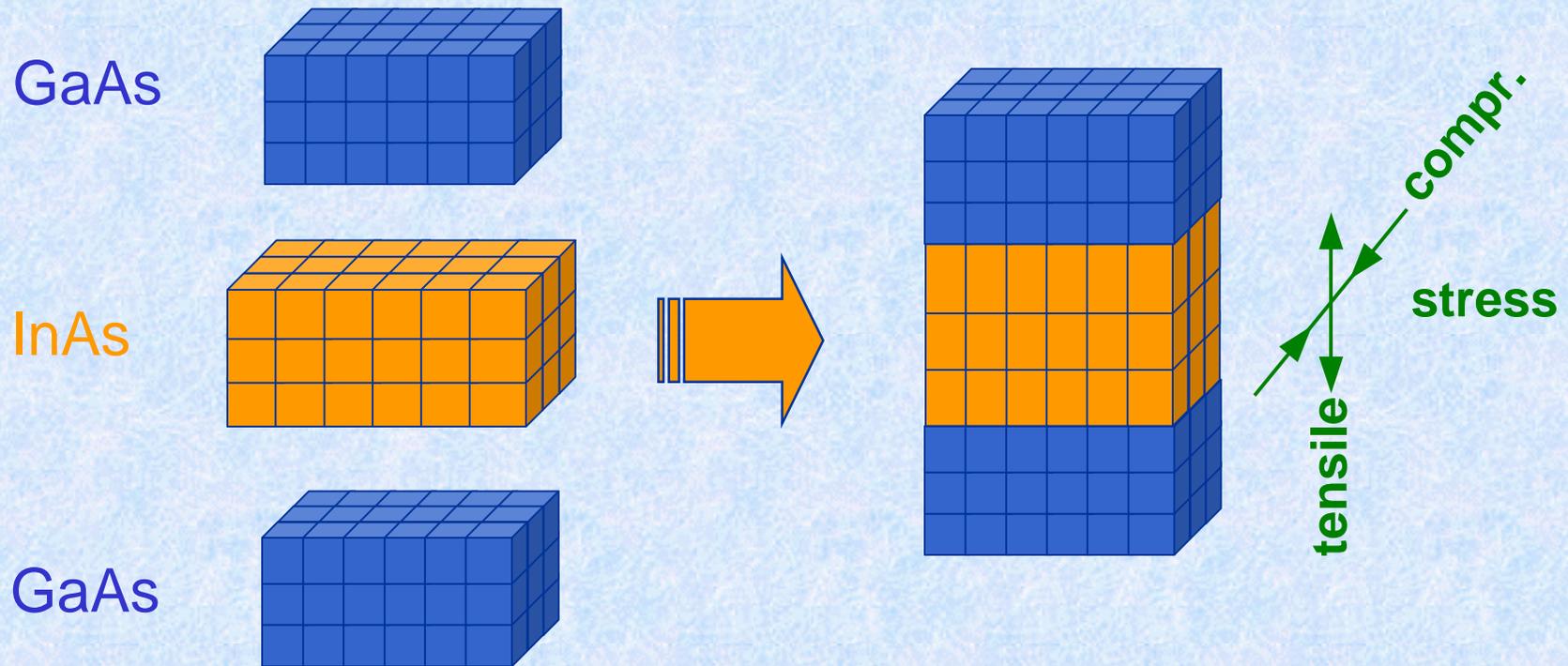
### Zero Growth

- Surface migration
- Interdiffusion

# Elastically Strained Material

Unstrained Material

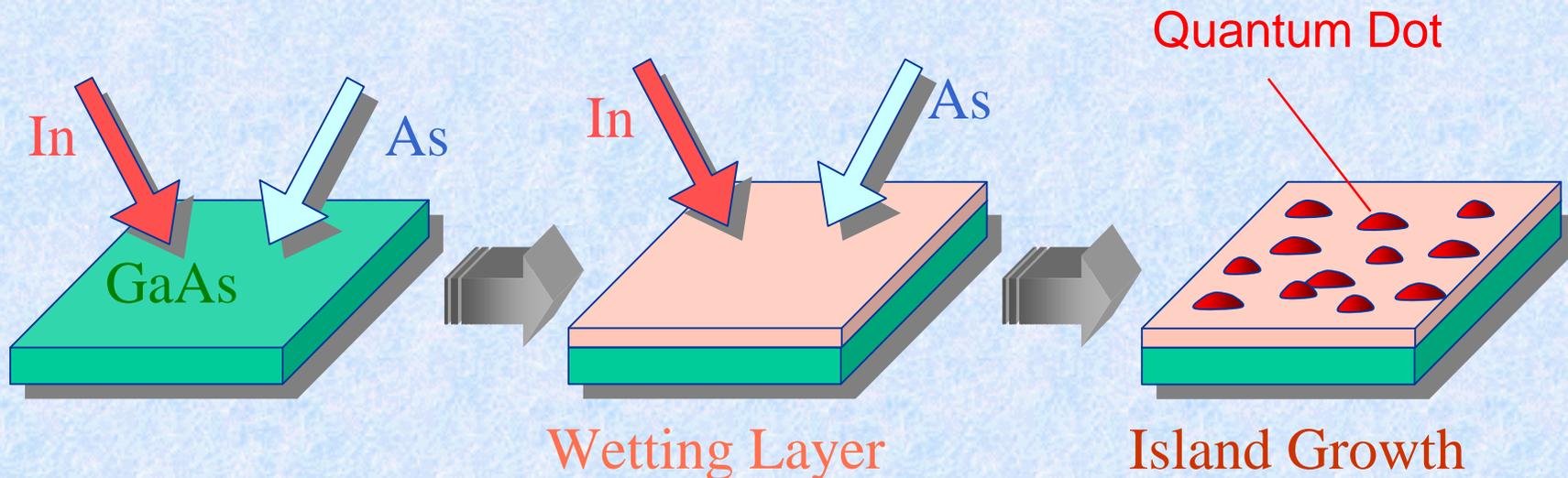
Pseudomorphic Growth



- InAs has 7.2 % larger lattice constant than GaAs
- For thin layers, growth without lattice defects possible

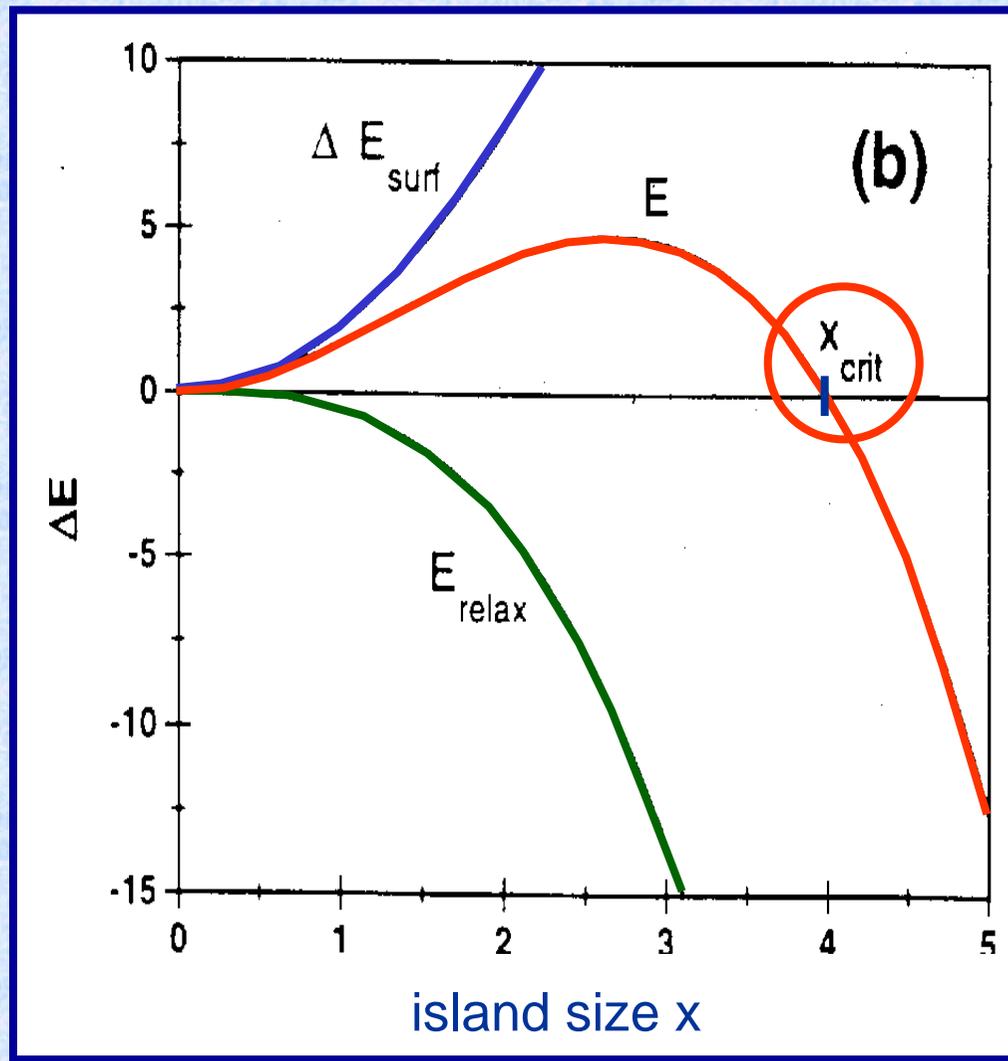
# Self-Assembled Quantum Dot Growth

Strain driven self-organisation effect  
(Stranski-Krastanov growth mode)



- Formation of atom-like islands ( $d \approx 10\text{-}20\text{ nm}$ ) due to energy minimation (wetting layer thickness: 1.7 ML for InAs)
- One dot contains still  $10^5$  atoms and behaves partially like bulk material (e.g., band structure properties)

# Dot Nucleation Process



Surface energy difference:

$$\Delta E_{\text{surf}} \sim x^2 \quad (\sim \text{surface})$$

Strain energy difference:

$$\Delta E_{\text{strain}} \sim x^3 \quad (\sim \text{volume})$$

Energy balance:

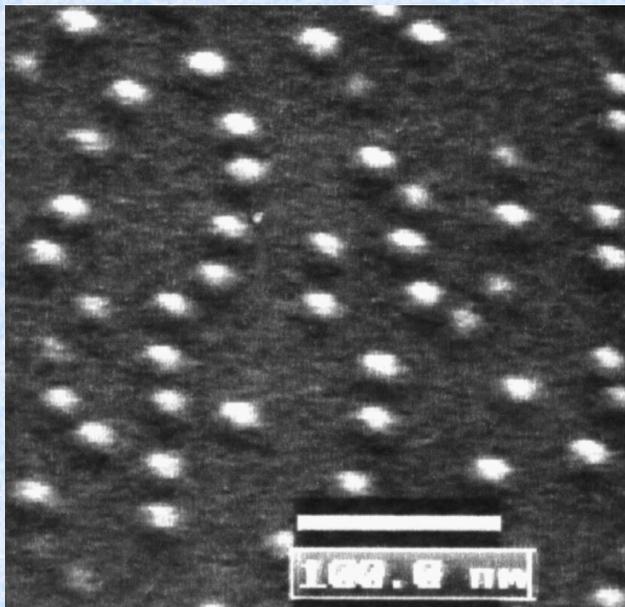
$$\Delta E = C\gamma x^2 - \kappa \varepsilon^2 C' x^3$$

$\gamma$  = surface energy,  
 $\kappa$  = elasticity module,  
 $\varepsilon$  = strain coefficient

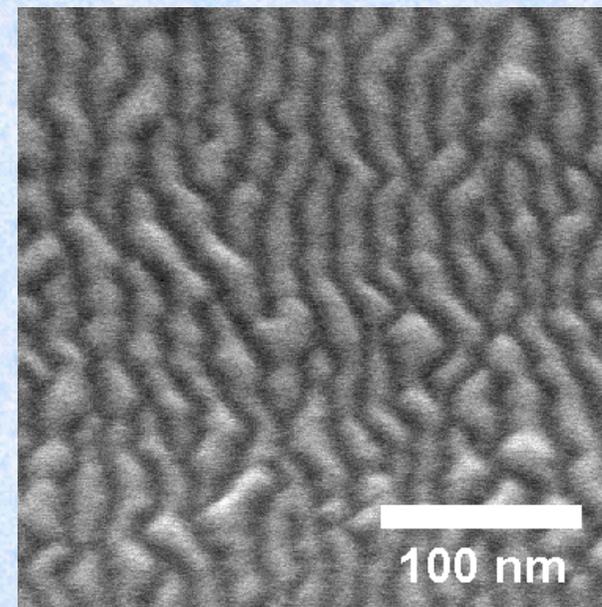
→ **island formation:**  
 $x > x_{\text{crit}}$

# InAs QD Formation on Different Substrates

## InAs Quantum Dots on GaAs



## InAs Quantum Dashes on AlGaInAs lattice matched to InP

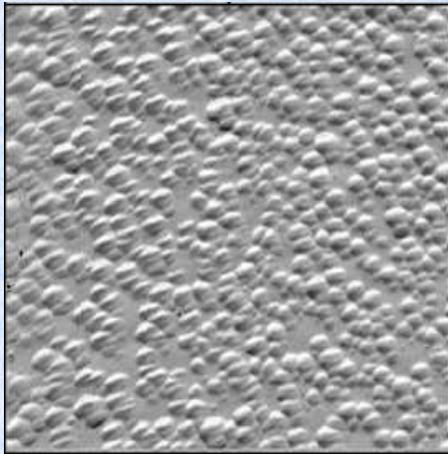


- QDash structures of high density are formed preferable in [0-11] crystal direction

# Dot density control by growth temperature

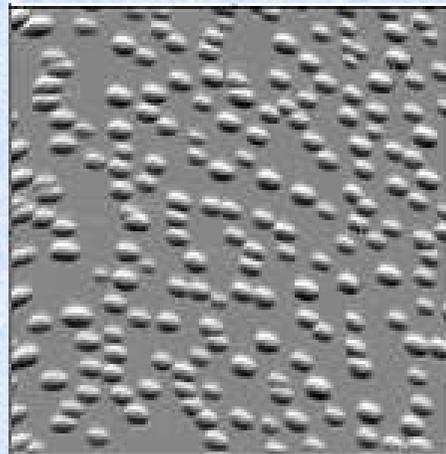
- $\text{In}_{0.6}\text{Ga}_{0.4}\text{As}$  QDs with fundamental transitions energy of around 1.3 eV
- AFM images of uncovered quantum dots on GaAs surfaces

$T_{\text{substrate}} = 480 \text{ }^\circ\text{C}$   
 $1 \times 1 \mu\text{m}$



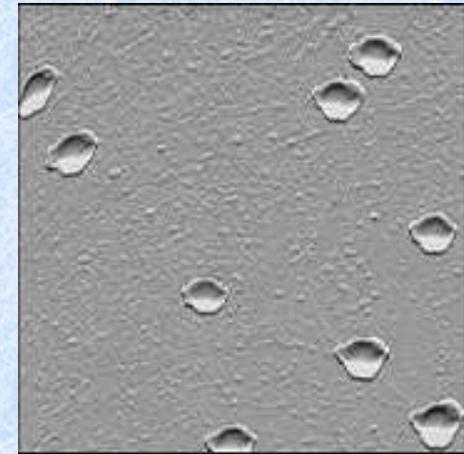
Dot density:  
 $6 \times 10^{10} \text{ cm}^{-2}$

$T_{\text{substrate}} = 510 \text{ }^\circ\text{C}$   
 $1 \times 1 \mu\text{m}$



Dot density:  
 $2 \times 10^{10} \text{ cm}^{-2}$

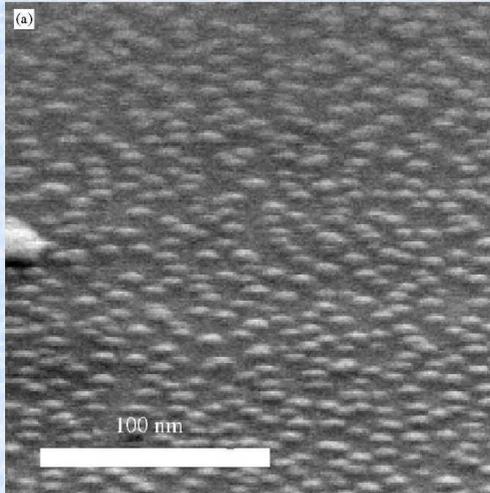
$T_{\text{substrate}} = 530 \text{ }^\circ\text{C}$   
 $1 \times 1 \mu\text{m}$



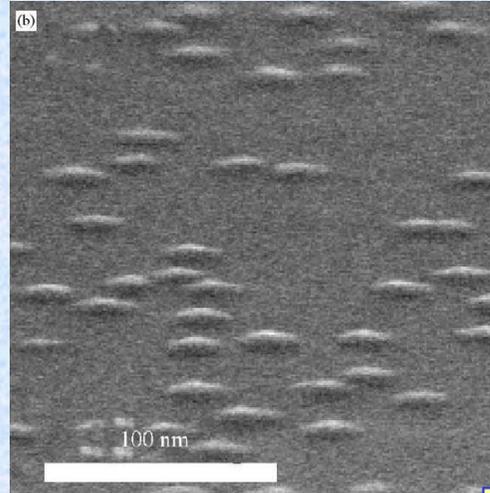
Dot density:  
 $< 10^9 \text{ cm}^{-2}$

# Influence of In Concentration on Dot Shape

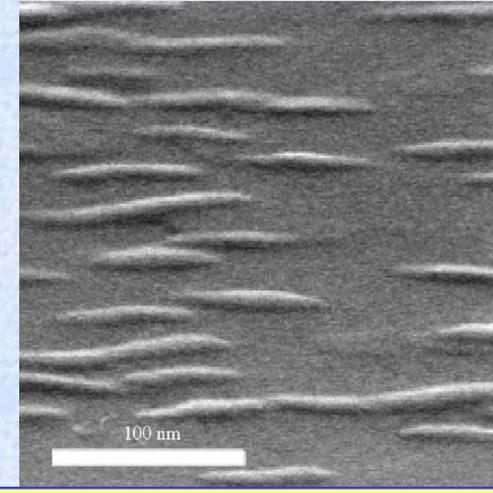
$x_{\text{In}} = 60\%$



$x_{\text{In}} = 45\%$

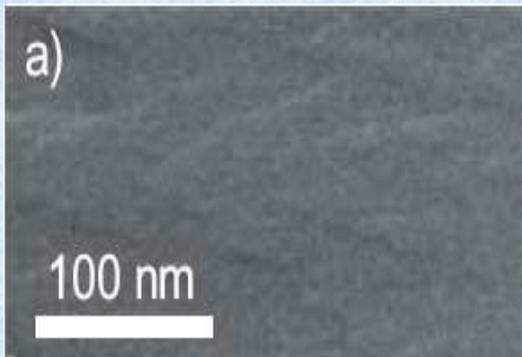


$x_{\text{In}} = 30\%$

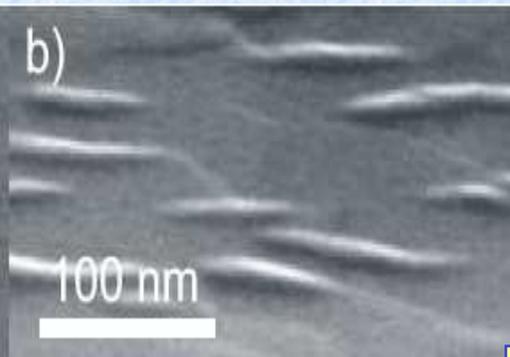


A. Löffler et al., JCG 286, 6 (2006)

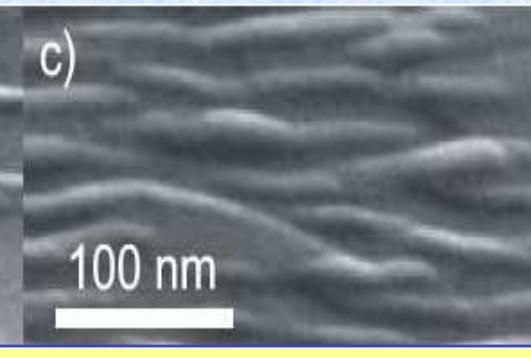
$x_{\text{In}} = 27\%$



$x_{\text{In}} = 30\%$

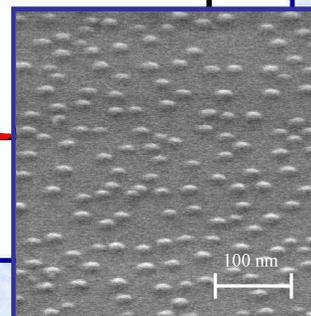
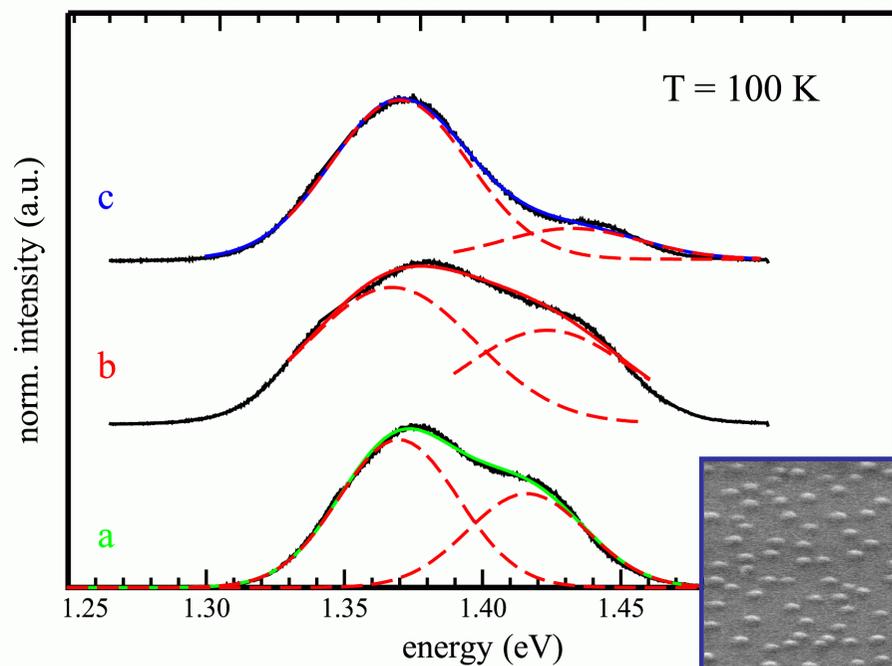
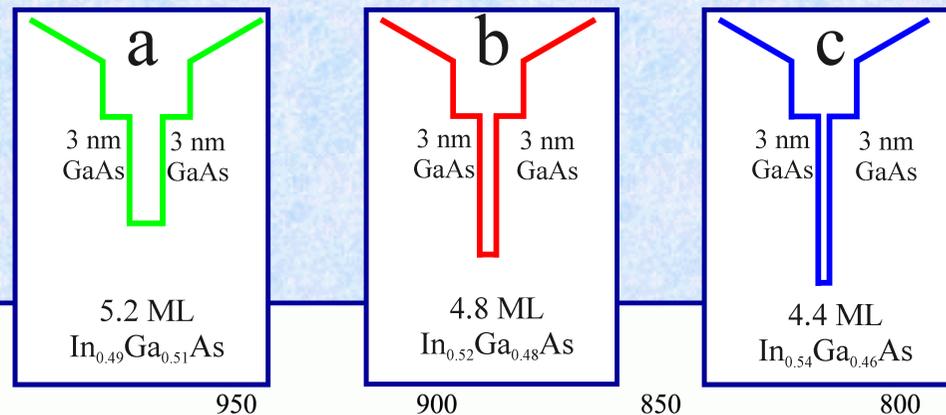


$x_{\text{In}} = 33\%$



A. Löffler et al., PSS C 3, 3815 (2006)

# Control of Dot Size by Layer Thickness



**Photoluminescence measurement of different QD samples at high excitation powers:**

energy separation between ground and first excited state increases with decreasing dot size:

a)  $\Delta x \approx 47$  meV

b)  $\Delta x \approx 56$  meV

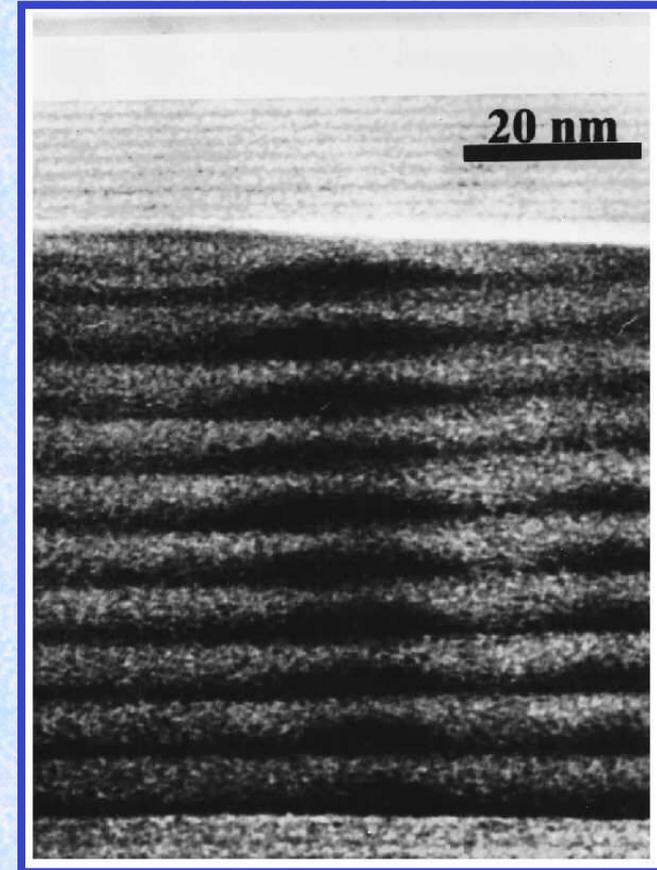
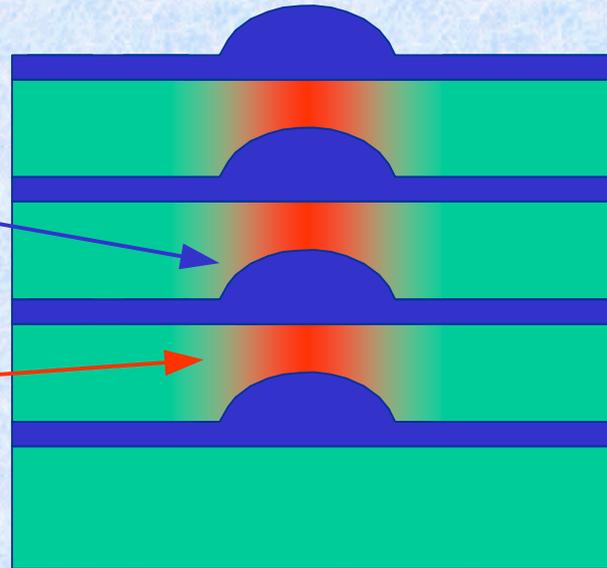
c)  $\Delta x \approx 65$  meV

⇒ **small QDs promising for tailoring gain spectrum of QD lasers**

J.P. Reithmaier et al.,  
phys. stat. sol. B 234, 3981 (2006)

# Strain Coupling

Preferential nucleation of InAs  
Aligned InAs Dot  
Local Strain  
InAs Dot  
GaAs layer

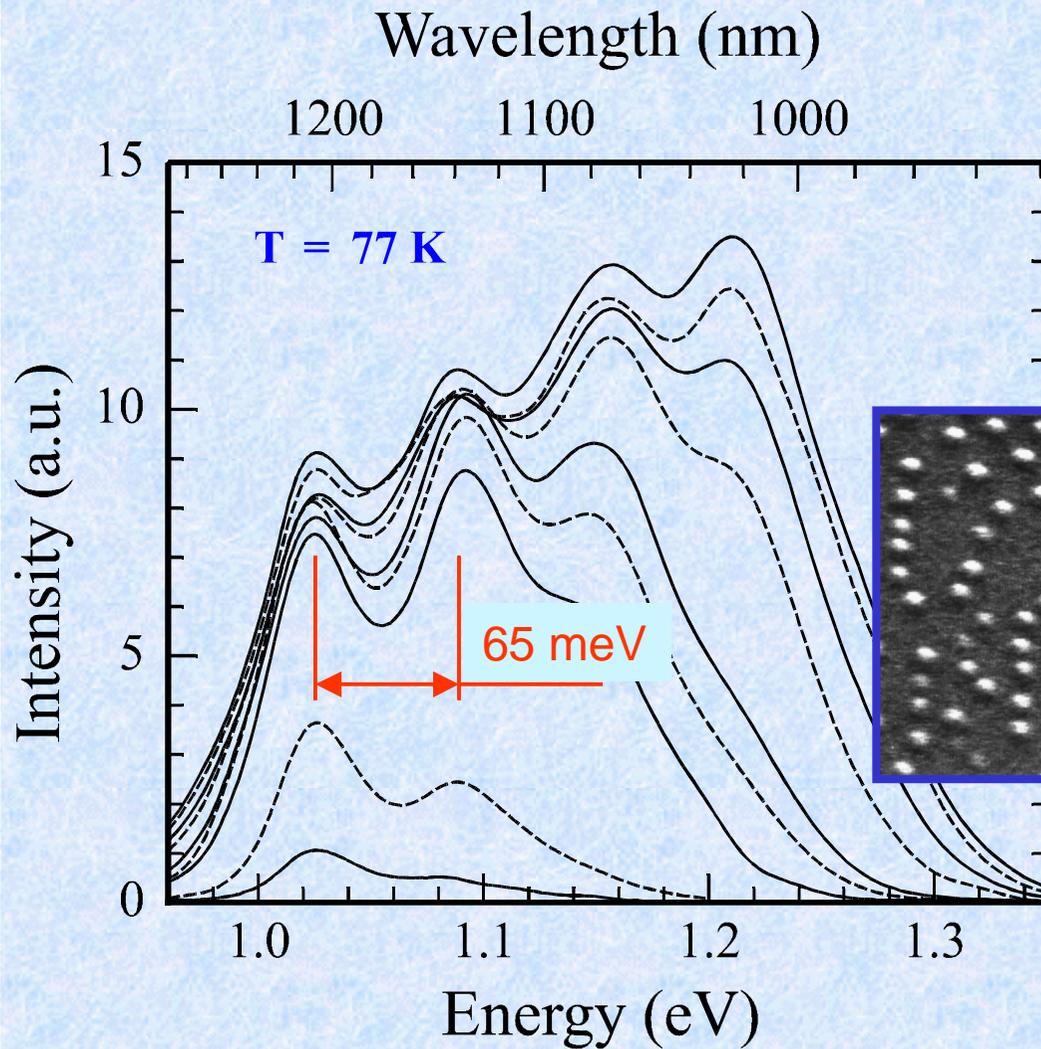


- Vertical alignment of QDs due to strain coupling
- Coupling strength dependent on thickness of GaAs intermediate layer ( $d < 30$  nm)

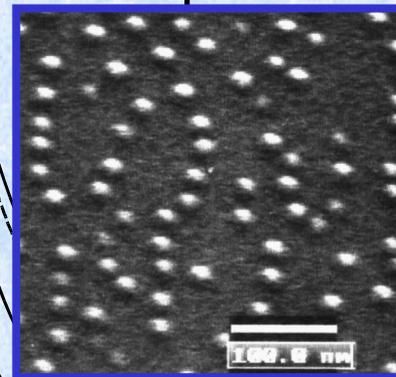
Maximov et al., J. Appl. Phys. 83, 5561 (1998)

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  - high power lasers (980 and 920 nm)
  - telecom lasers and amplifiers (1.3 and 1.55  $\mu\text{m}$ )
  - single photon emitters and microcavity lasers

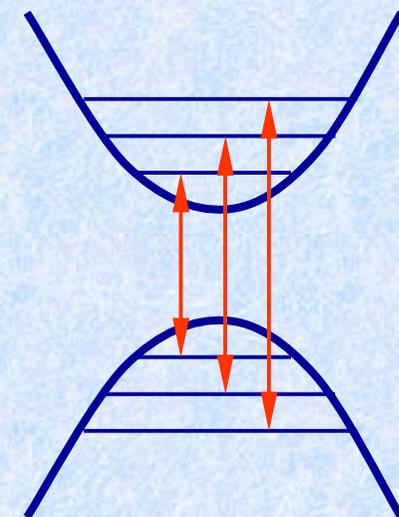
# Potential shape of SA-QDots



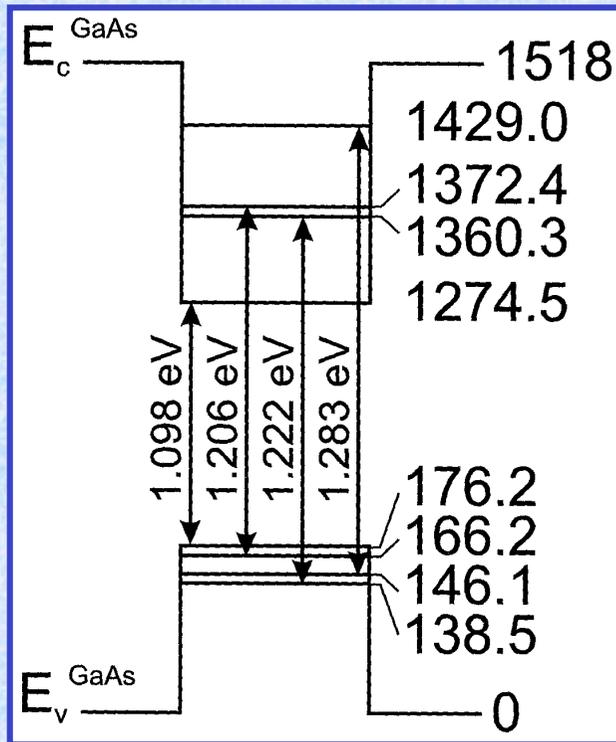
- Electroluminescence at 77 K
- Nearly equidistant transition energies



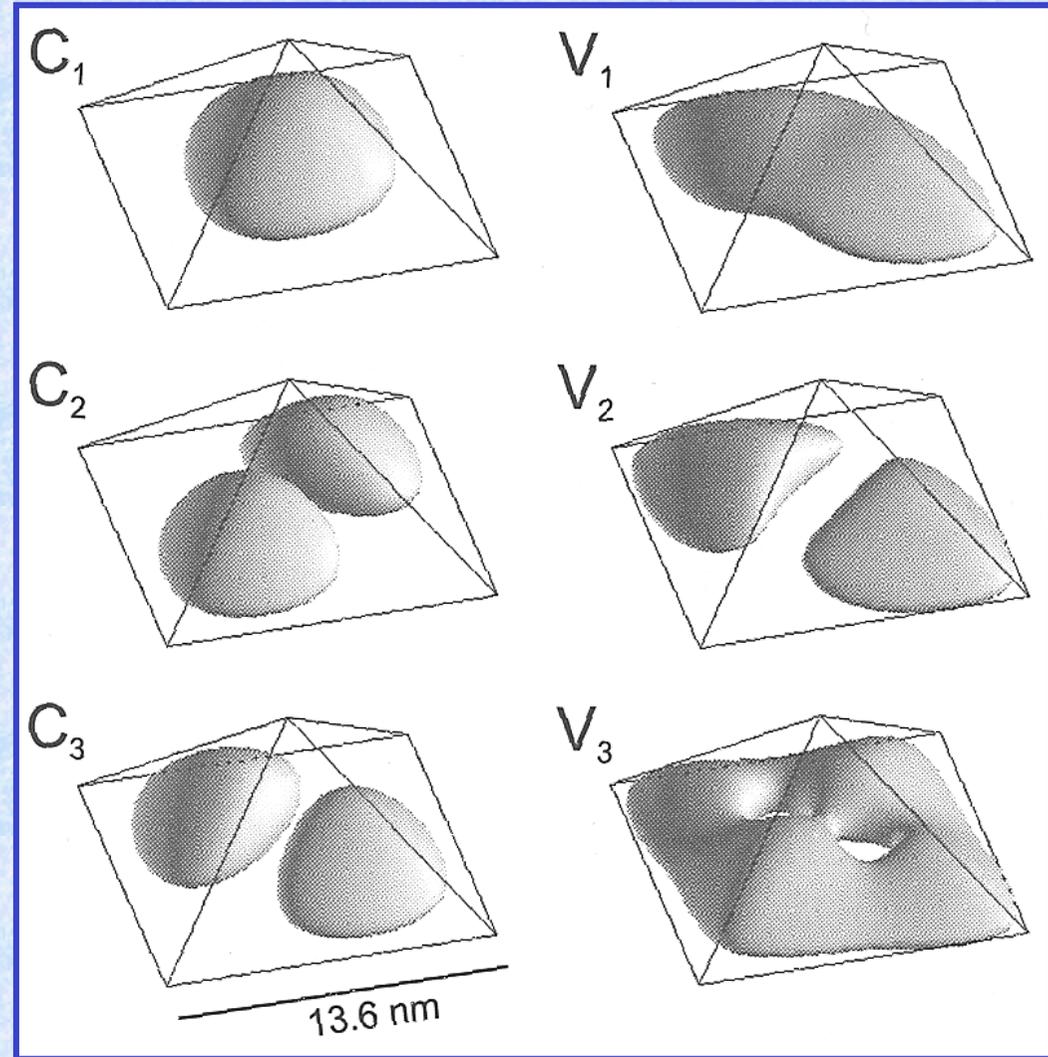
→ Parabolic potential shape



# Quantum Dot Wavefunctions



s and p type electron and hole states for an InAs/GaAs quantum dot (exciton formation is not taken into account).

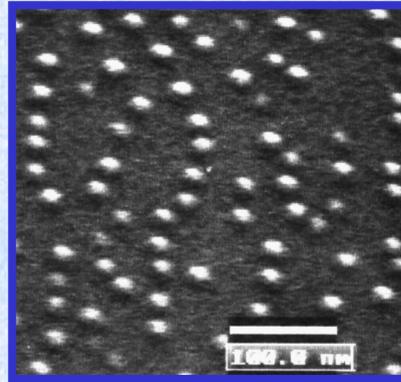
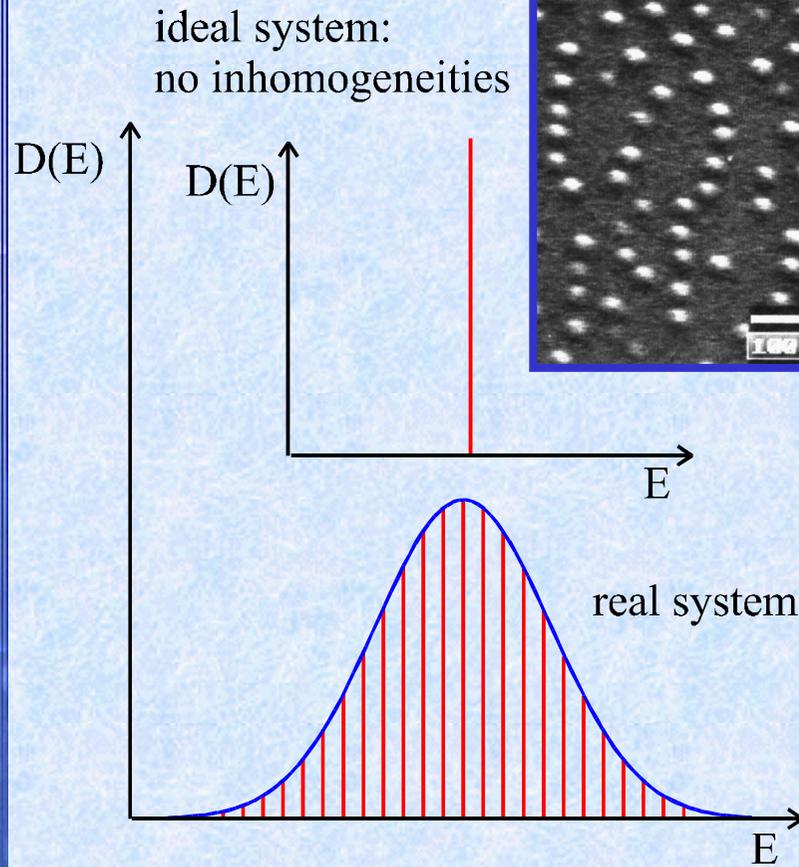


electron states

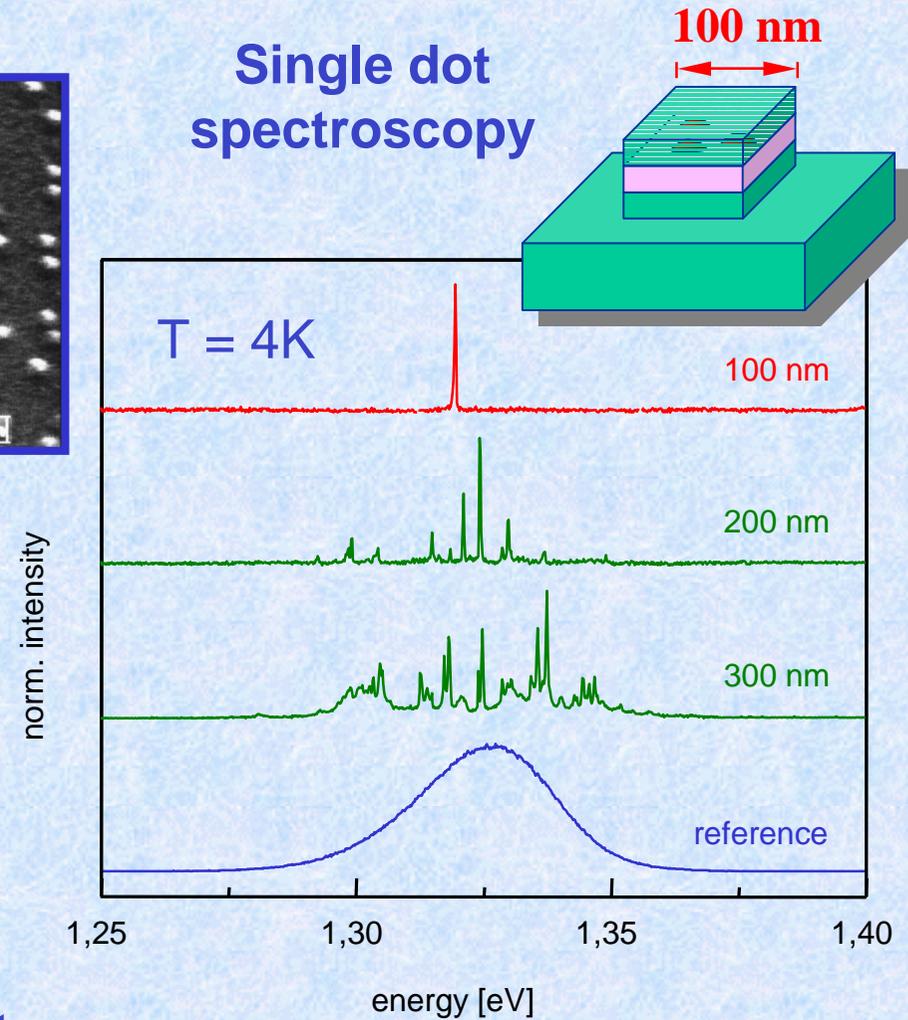
heavy hole states

D. Bimberg et al., "Quantum Dot Heterostructures", Wiley, 1999

# Self-Assembled Quantum Dots



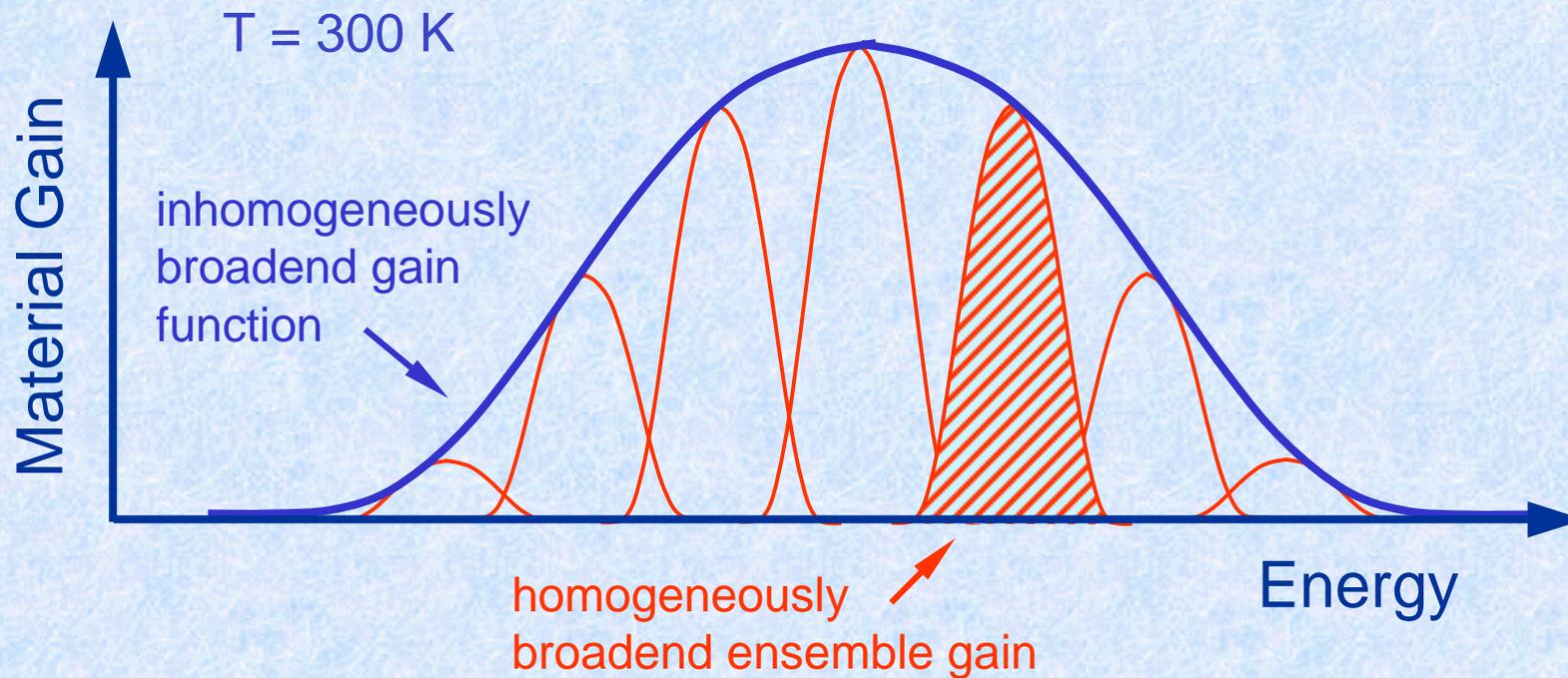
## Single dot spectroscopy



- Broad emissions spectrum of a dot ensemble due to size fluctuations

- Single line by dot selection

# Spectrally Distributed Gain

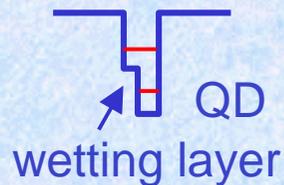


⇒ multi-wavelength amplification due to weak overlap between gain functions of different dot ensembles

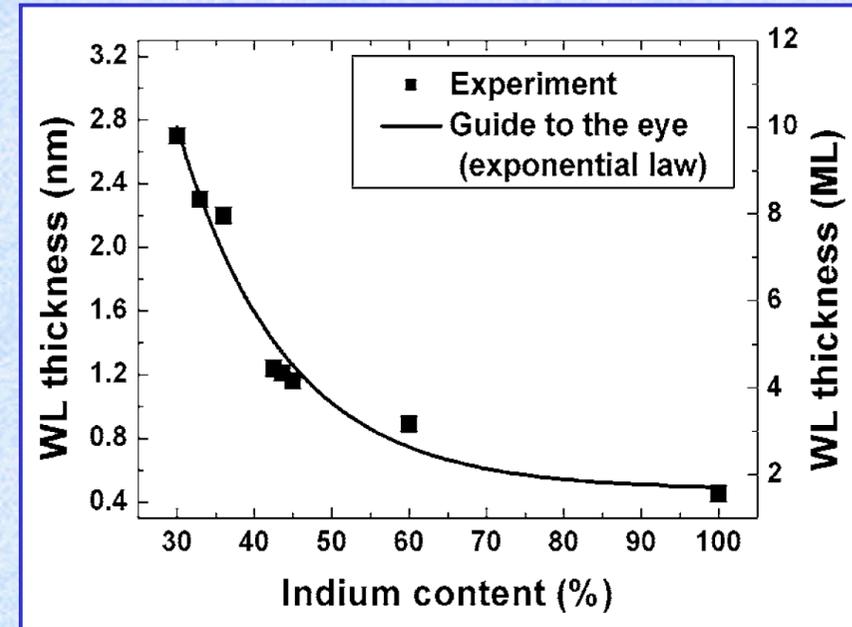
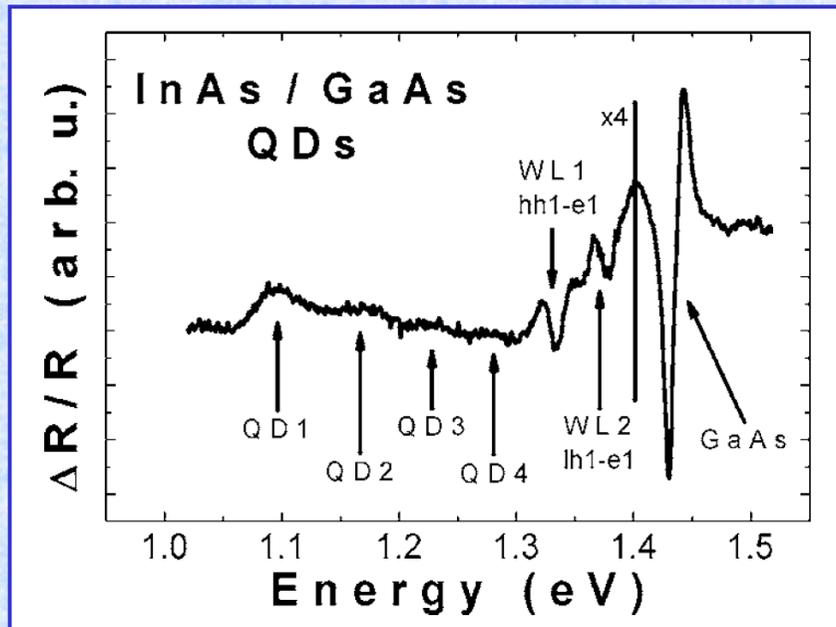
homogeneous linewidth  $\approx 5 - 10$  meV (RT)

inhomogeneous linewidth  $\approx 30 - 50$  meV

# Wetting layer thickness



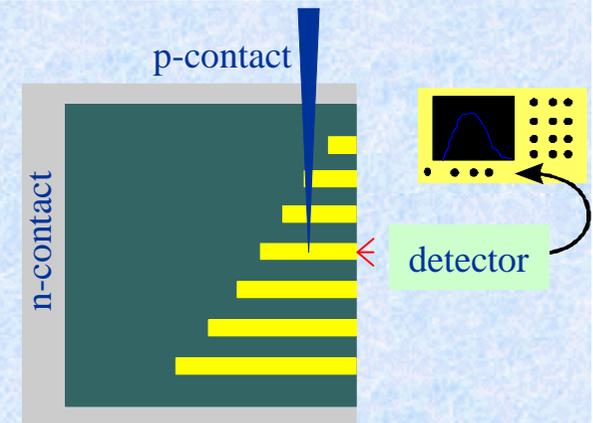
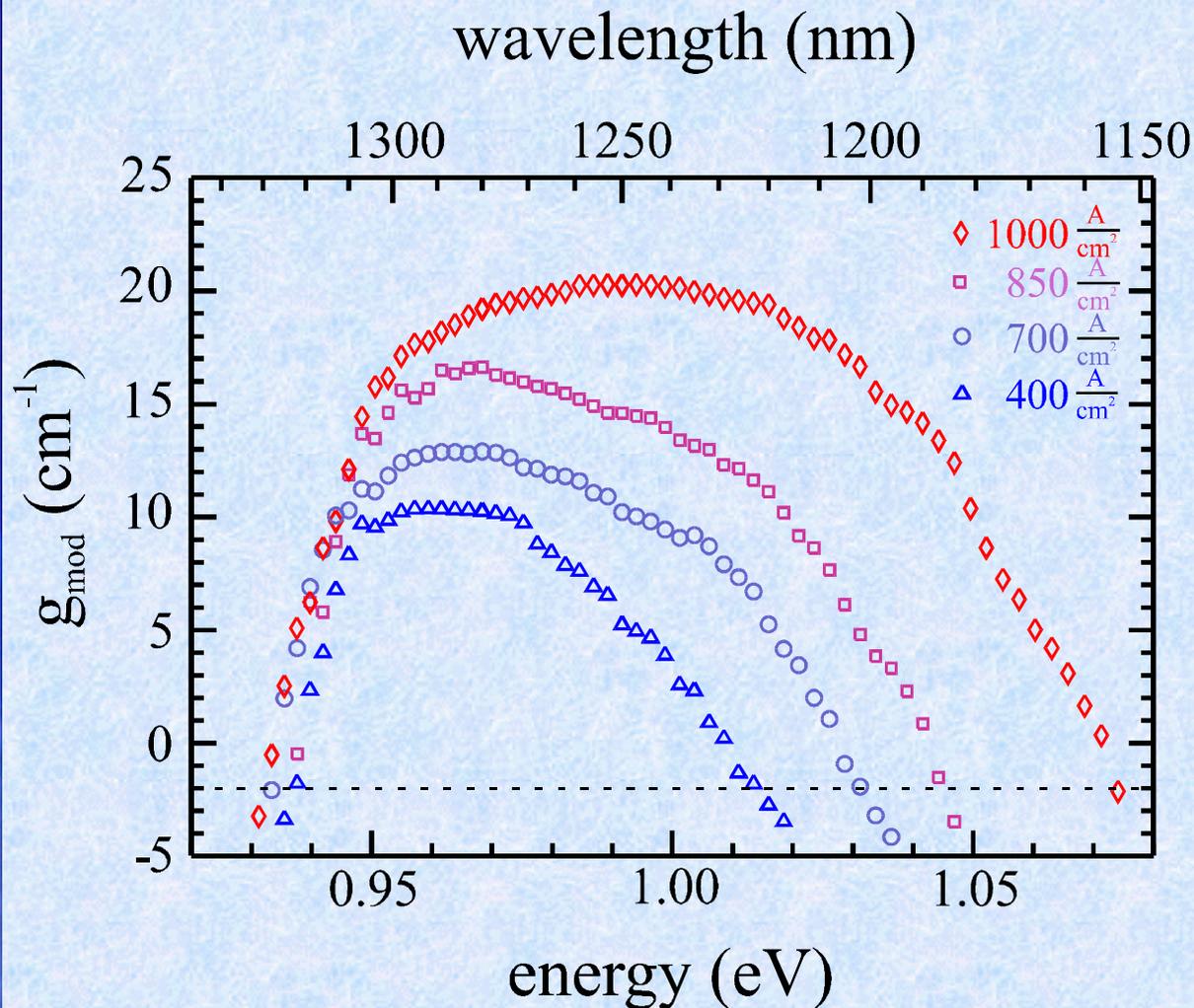
## Photoreflectance spectroscopy



- Reflectance spectroscopy as very sensitive tool for optical transitions
- WL transitions clearly visible as  $hh_1-e_1$  and  $lh_1-e_1$  transitions
- WL transitions dependent on In concentration and thickness
- WL thickness varies from 1.6 ML (InAs) up to about 10 MLs ( $\text{In}_{0.3}\text{Ga}_{0.7}\text{As}$ )

G. Sek et al., JAP 100, 103529 (2006)

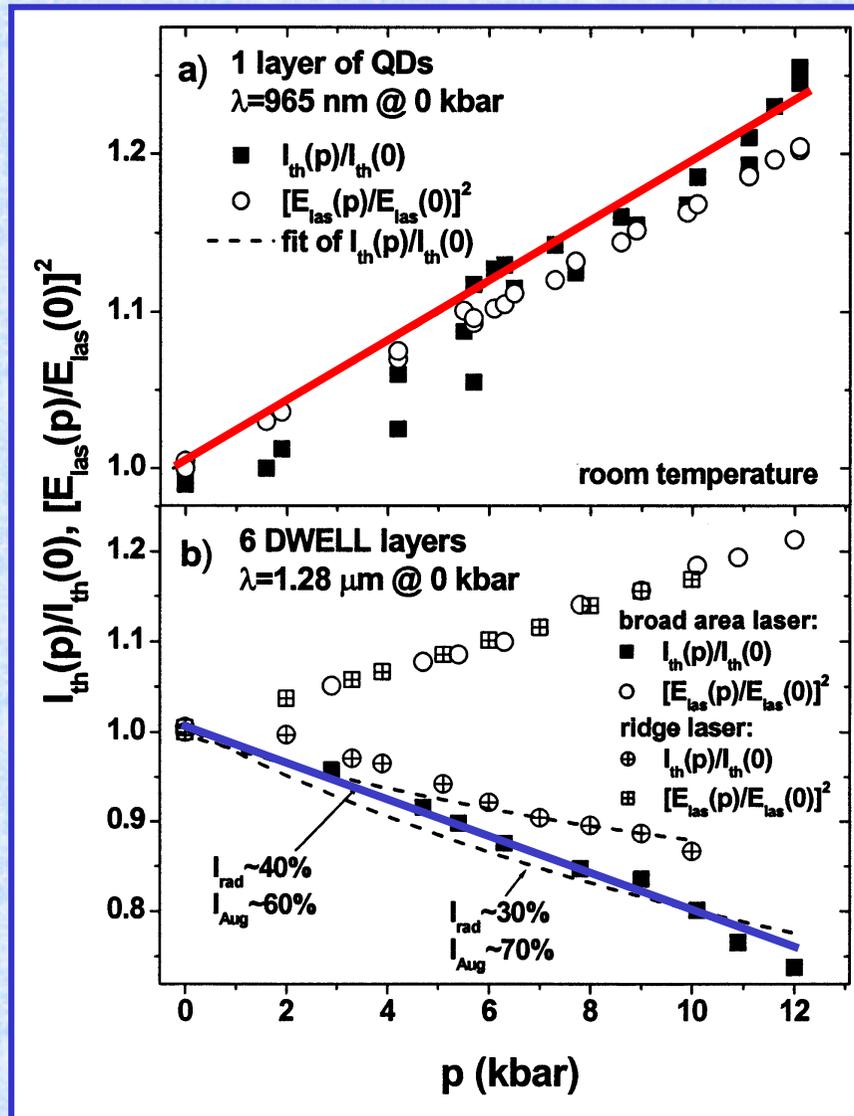
# Gain Saturation and Flat Gain Profile



- High gain at transition energy already at low current density
- Fundamental transition saturates and higher order transitions contribute to the gain
- maximum gain of 17 - 20  $\text{cm}^{-1}$  (6 dot layers stack)

→ surprisingly low modal gain of 3  $\text{cm}^{-1}$  per dot layer

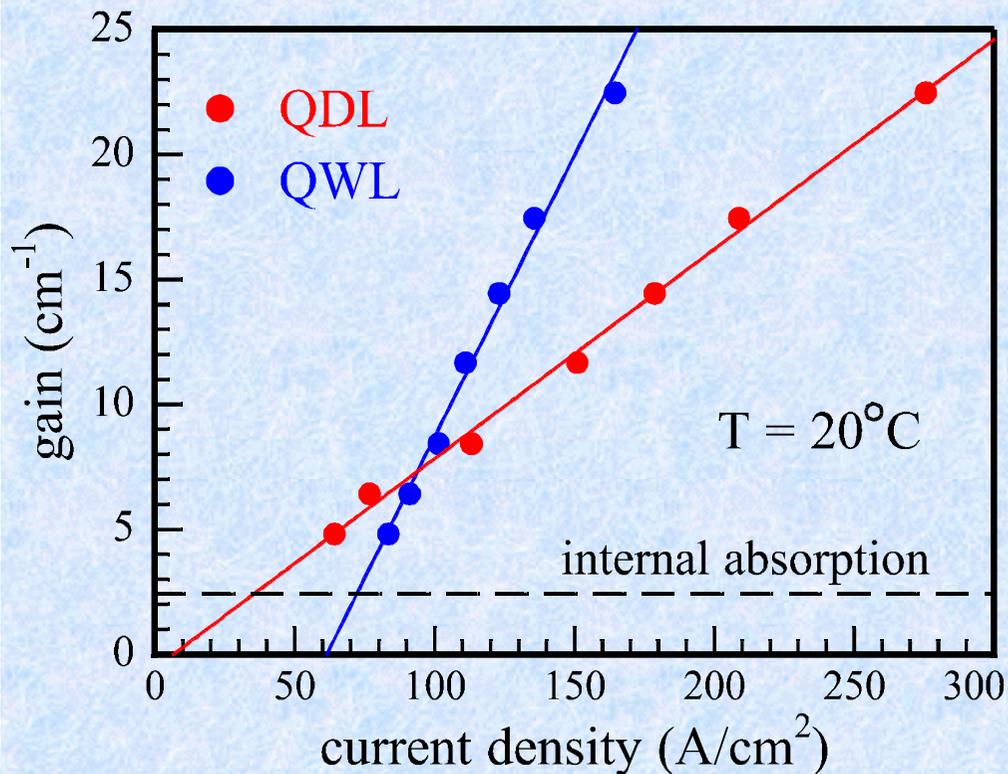
# Auger Effect in Different Types of QDs



- Hydrostatic pressure measurements show very different behavior of 980 nm and 1.3  $\mu\text{m}$  QD lasers.
  - $I_{th}(980 \text{ nm})$  increases with  $p$  and  $E_{las}$  with  $p^2$  as expected for radiative recombination
  - $I_{th}(1.3 \text{ } \mu\text{m})$  decreases by 26 % with  $p$  while  $E_{las}$  increases.
- Behavior for 1.3  $\mu\text{m}$  could be most likely explained by a significant contribution of **Auger recombination** as non-radiative process while 980 nm QD lasers seems to be dominated by **radiative recombination**.

I.P. Marco et al., JSTQE 9, 1300 (2003)

# Gain of 980 nm QD and QW-Laser



- Inversion condition already achieved at lower carrier densities  
→  $j_{tr} = 36 \text{ A/cm}^2$  ( $\alpha_i = 2.2 \text{ cm}^{-1}$ )  
 $j_{th} = 54 \text{ A/cm}^2$  (2mm,HR/HR)
- Modal gain for single dot layer limited

$$\text{Threshold gain : } g_{th} = \alpha_i + \frac{1}{L} \ln\left(\frac{1}{\sqrt{R_1 R_2}}\right)$$

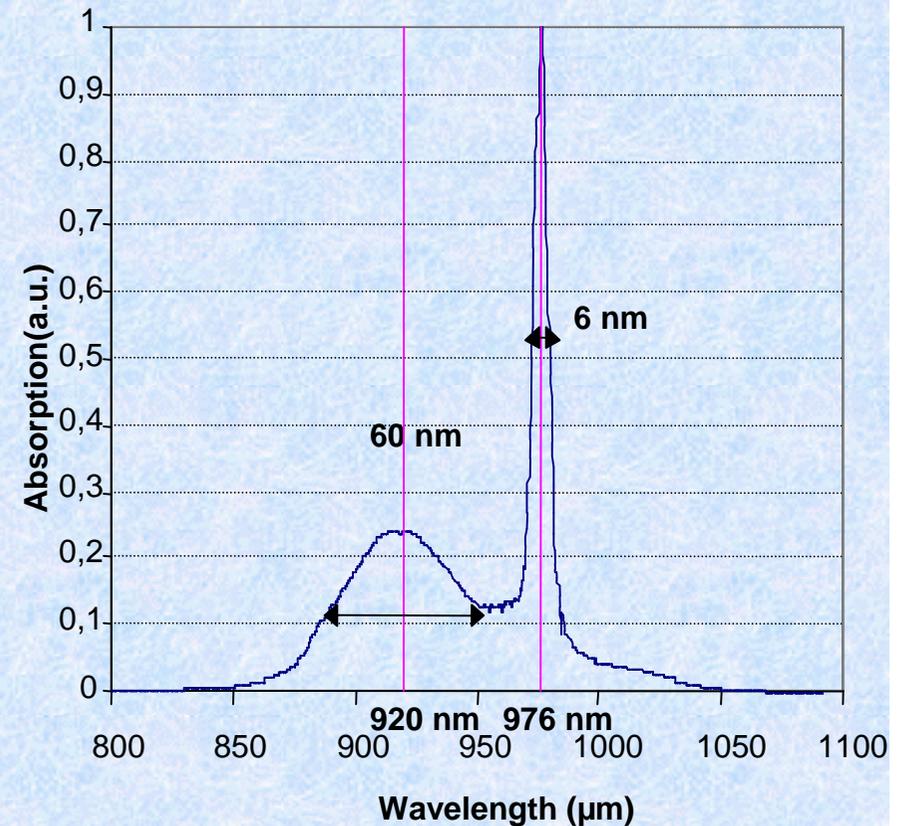
F. Klopff et al., APL 77, 1419 (2000)

- Introduction and Some Basic Theory
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- Application examples of QD lasers
  - high power lasers (980 / 920 nm)
  - ultra-broad band lasers (1.55  $\mu\text{m}$ )
  - ultra-fast broadband SOAs (1.55  $\mu\text{m}$ )

# Motivation for Uncooled Pump Lasers

- Major costs in high power pump modules: **Peltier cooler**
- External wall plug efficiency of pump module dominated by Peltier cooler consumption.
- **Strong cost reduction possible by passive cooling**
- Passive cooling with QW laser impossible due to wavelength shift ( $> 20$  nm between 0 – 65 °C)
- 920 nm favourable due to broader absorption band
- **High power 920 nm QD laser with temperature shift  $< 10$  nm possible**

Absorption spectrum of Yb fibre



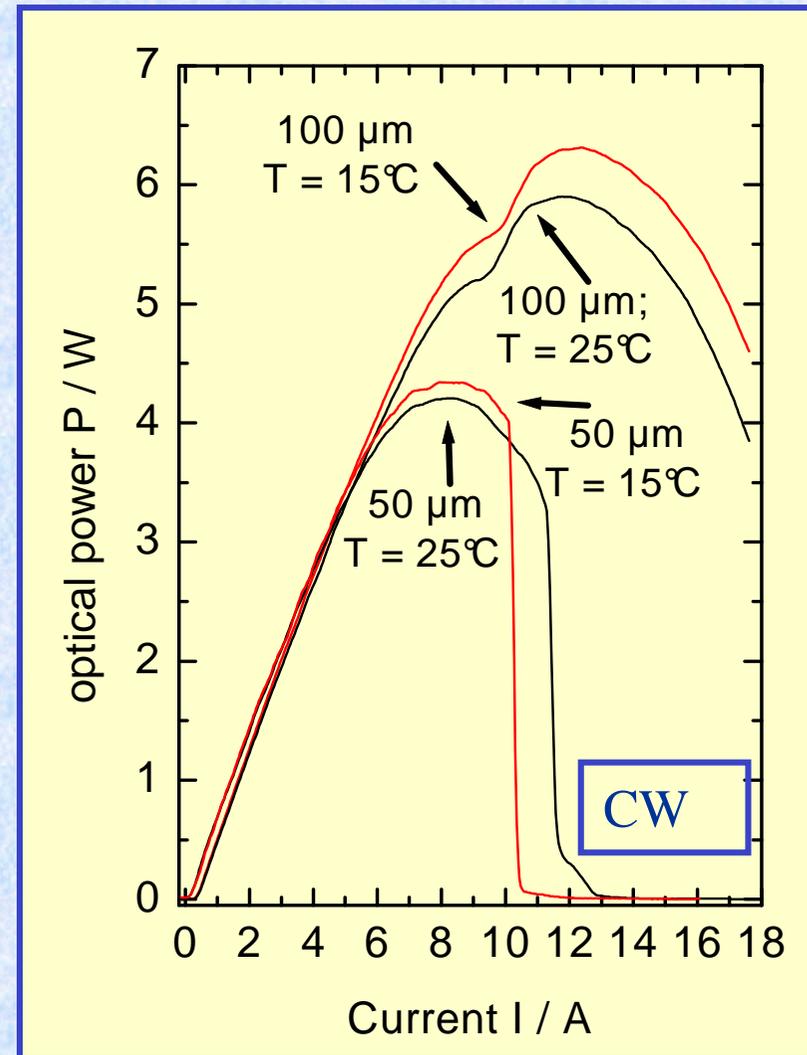
measured by Alcatel

# High Power CW-Operation (980 nm laser)

**FBH**

Ferdinand-Braun-Institut  
für Höchstfrequenztechnik

- Epi-side down mounted devices mounted on heat sink
- $P_{\max} = 4.3 \text{ W (6.3 W)}$  for 50  $\mu\text{m}$  (100  $\mu\text{m}$ ) stripe width in cw
- Record in cw output power and power per stripe width (**86 mW/ $\mu\text{m}$** )
- $P_{\max} = 5.4 \text{ W (9.5 W)}$  for 50  $\mu\text{m}$  (100  $\mu\text{m}$ ) stripe width in qcw (100  $\mu\text{s}$  pulses)  
→ (**104 mW/ $\mu\text{m}$** )



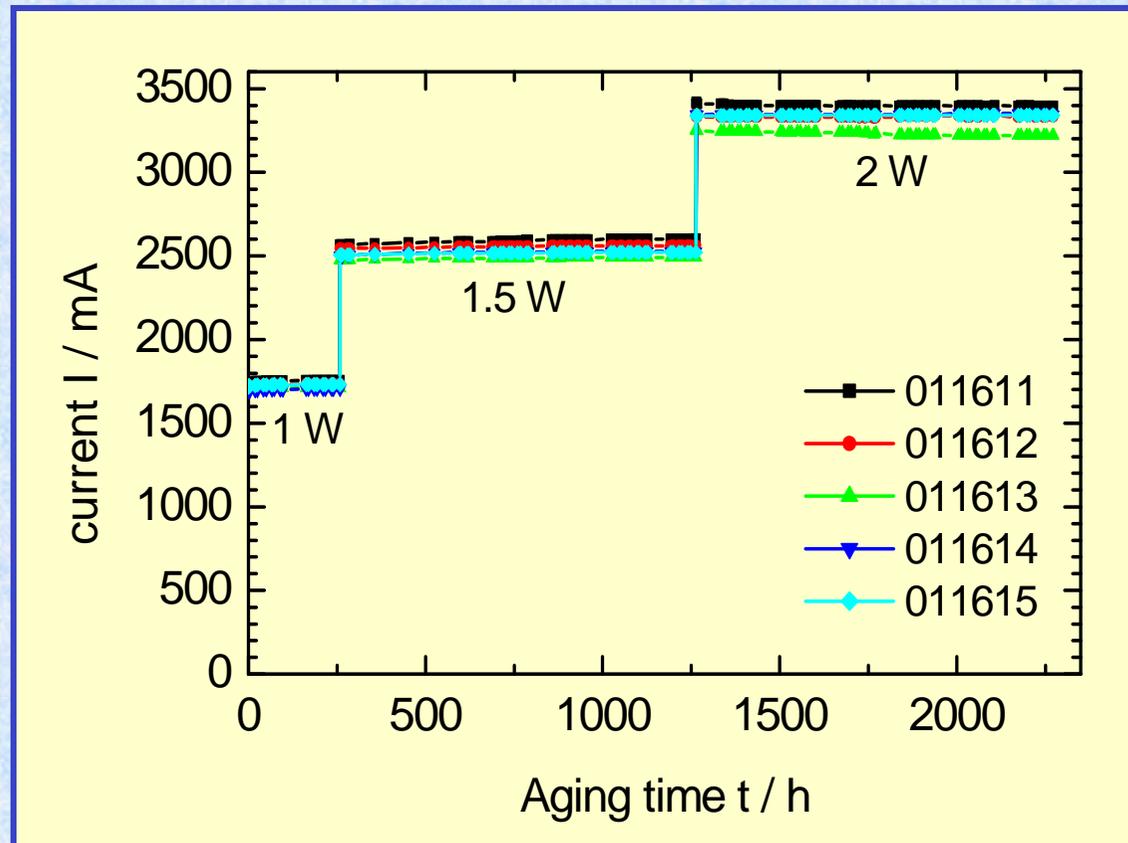
B. Sumpf et al., EL 39, 1655 (2003)

# High Power Lifetime Measurements (980 nm)

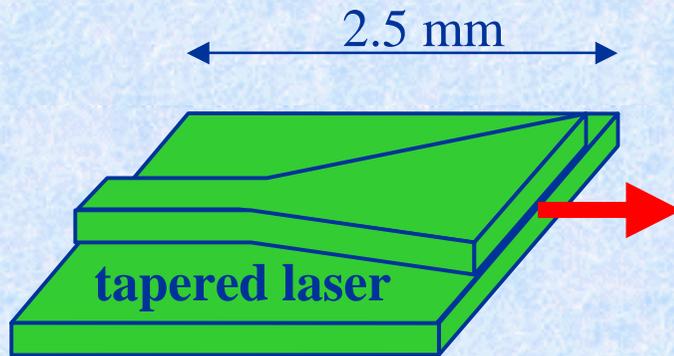
**FBH**

Ferdinand-Braun-Institut  
für Höchstfrequenztechnik

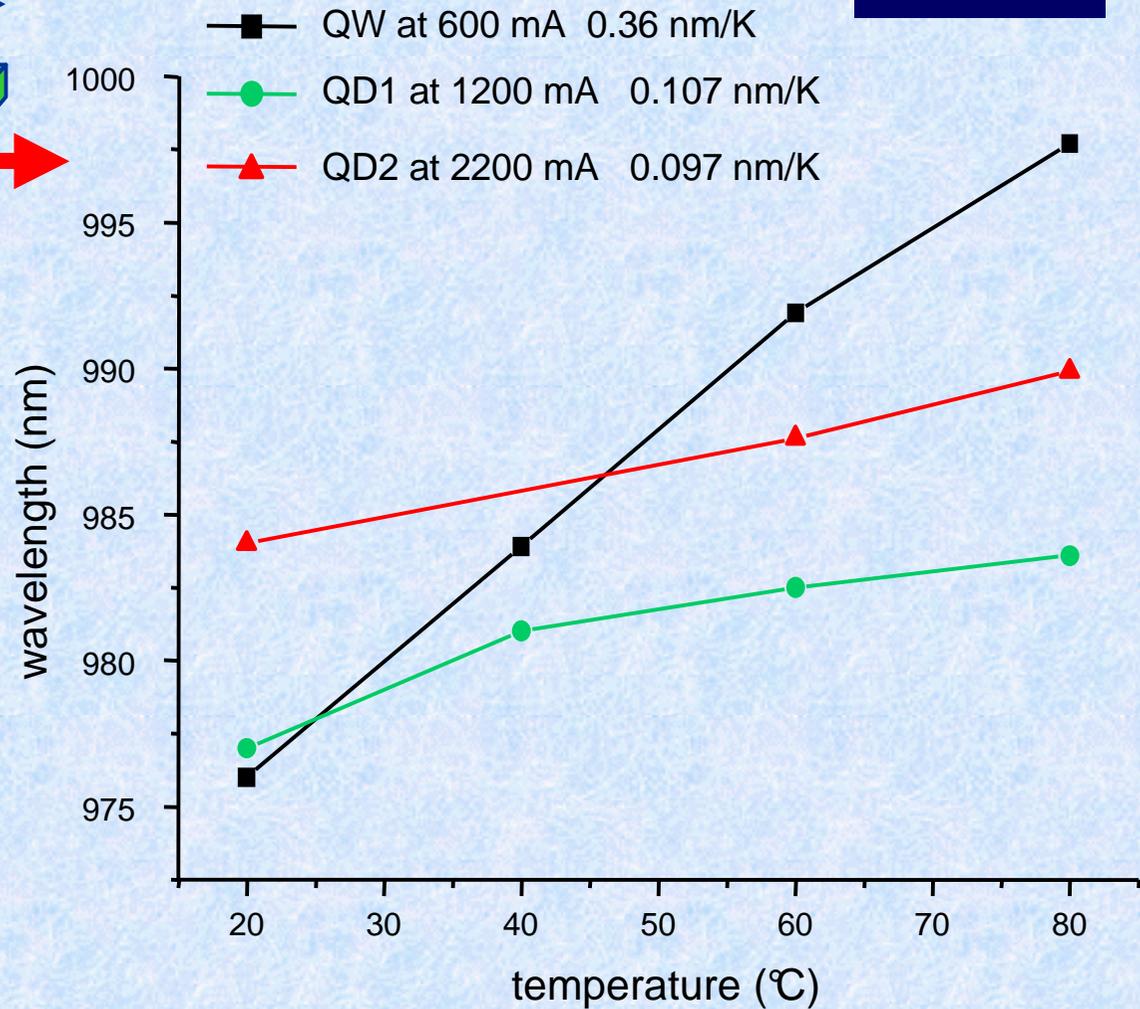
- Epi-side down mounted devices
- 1 mm long, 70  $\mu\text{m}$  wide stripes
- 2300 h without failures
- > 1000 h at 2 W ( $\approx 30 \text{ mW}/\mu\text{m}$ )



# Temperature Stability of Wavelength (980 nm)

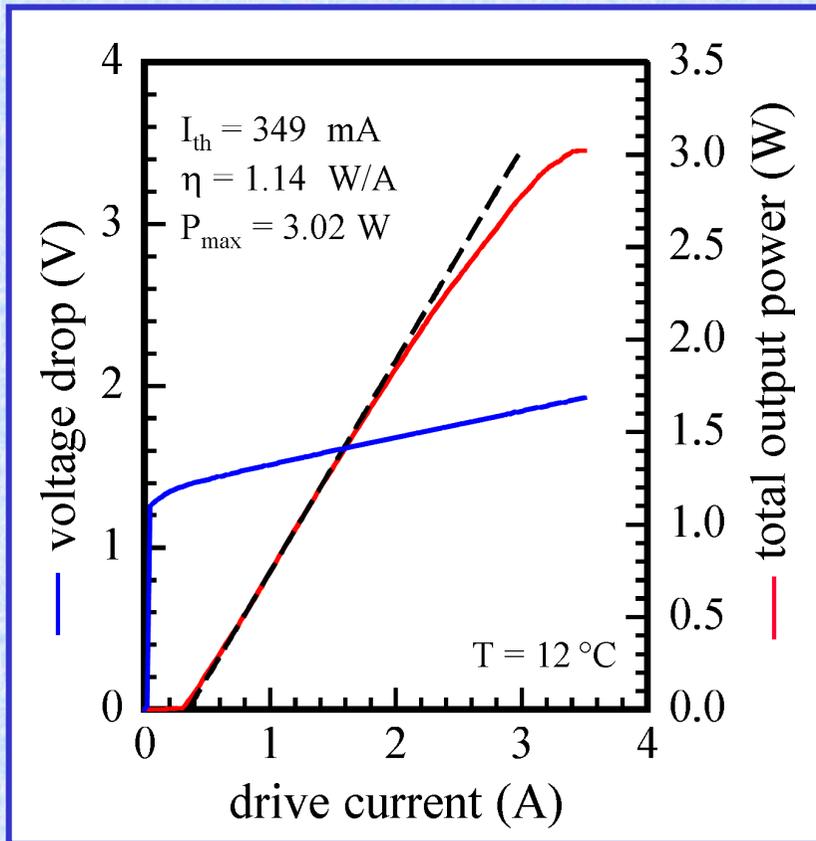


- Very low temperature dependence for QD lasers
- Between 20 – 80°C  
QW:  $\Delta\lambda = 22 \text{ nm}$   
QD:  $\Delta\lambda < 6 \text{ nm}$

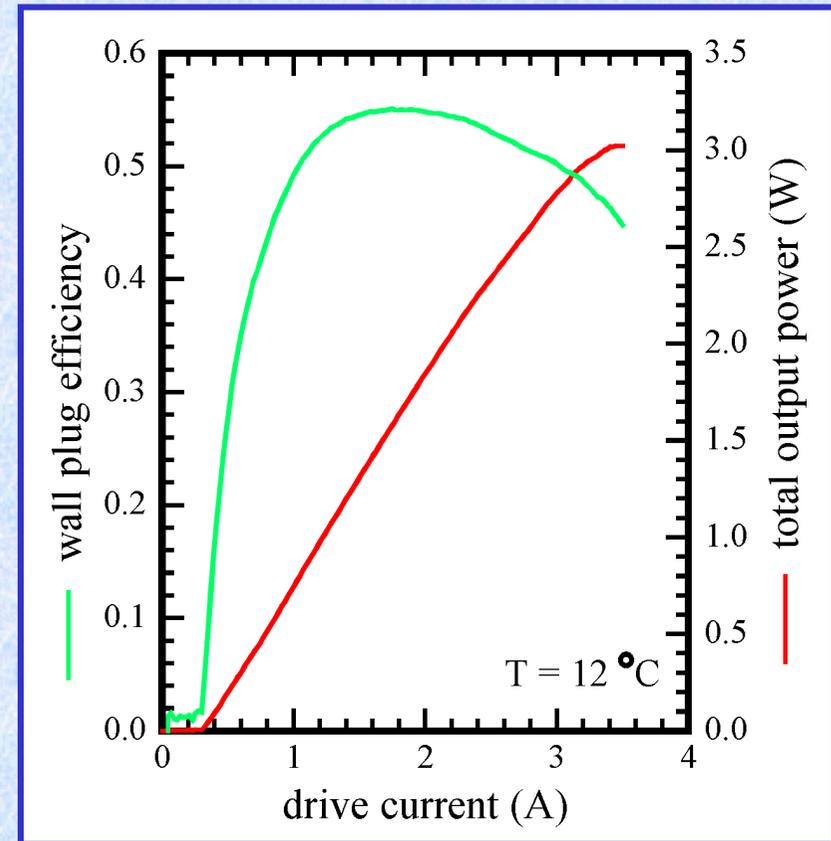


S.C. Auzanneau et al., APL 84, 2238 (2004)

# High Power Measurement Data of 920 nm QD Lasers



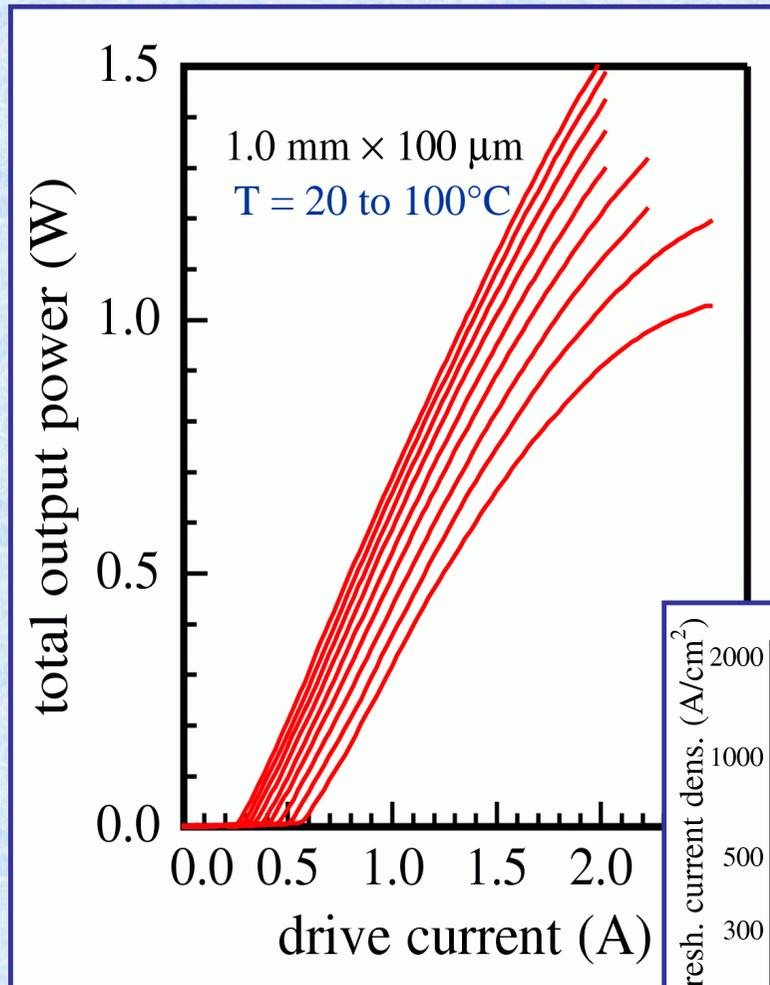
As-cleaved device,  
1 mm long, 100  $\mu\text{m}$  wide



Wall-plug efficiency of 55 % at 1.5 W

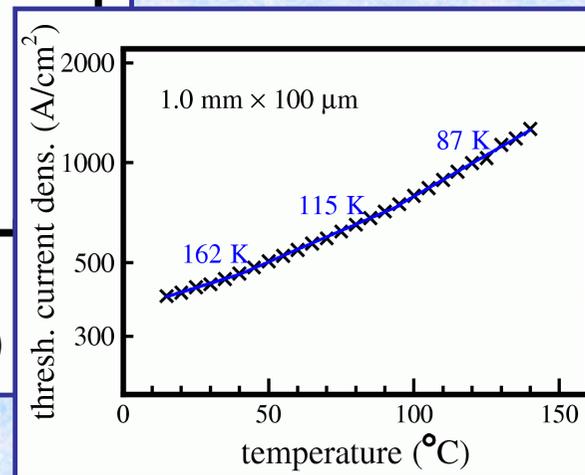
S. Deubert et al, EL 41, 1125 (2005)

# Temperature Stability of Laser Performance (920 nm)



## CW operation of a 1.0 mm × 100 μm:

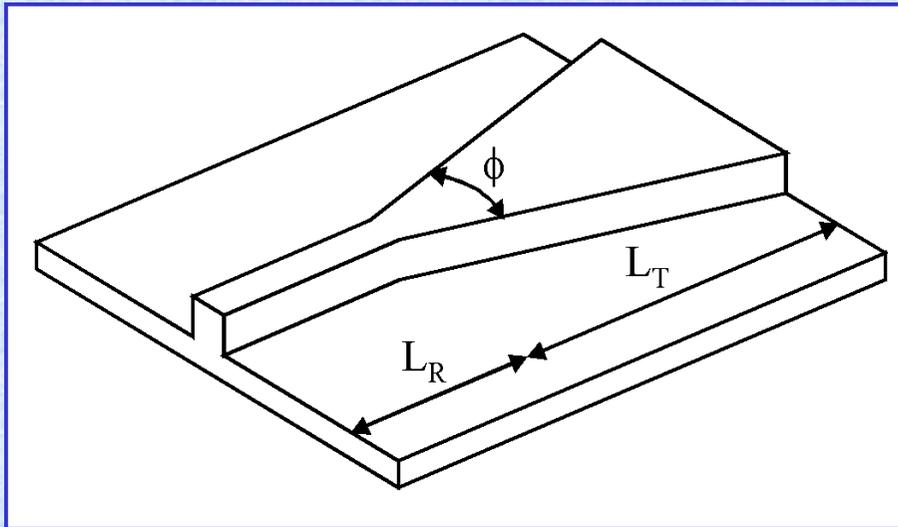
- Temperature from 20 to 100°C in steps of 10°C
- High temperature stability of threshold and efficiency
- Emission wavelength at about 915 nm
- **Output power of  $P > 1 \text{ W}$  @ 100°C**



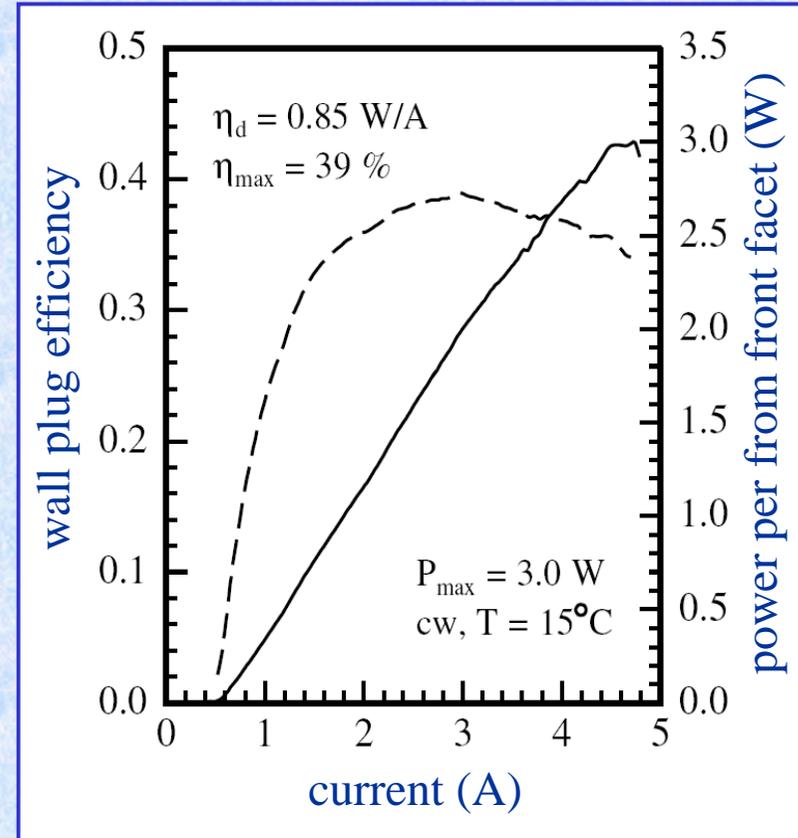
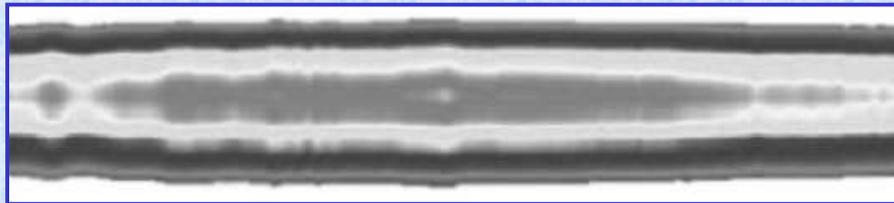
## Pulsed operation

- $T_{\text{max}} > 140 \text{ }^{\circ}\text{C}$
- High  $T_0$  of 162 K up to 40 °C and  **$> 150\text{K}$  up to 60°C**

# 920 nm Tapered QD Lasers



Near field intensity profile at  $P_{cw} = 1 \text{ W}$

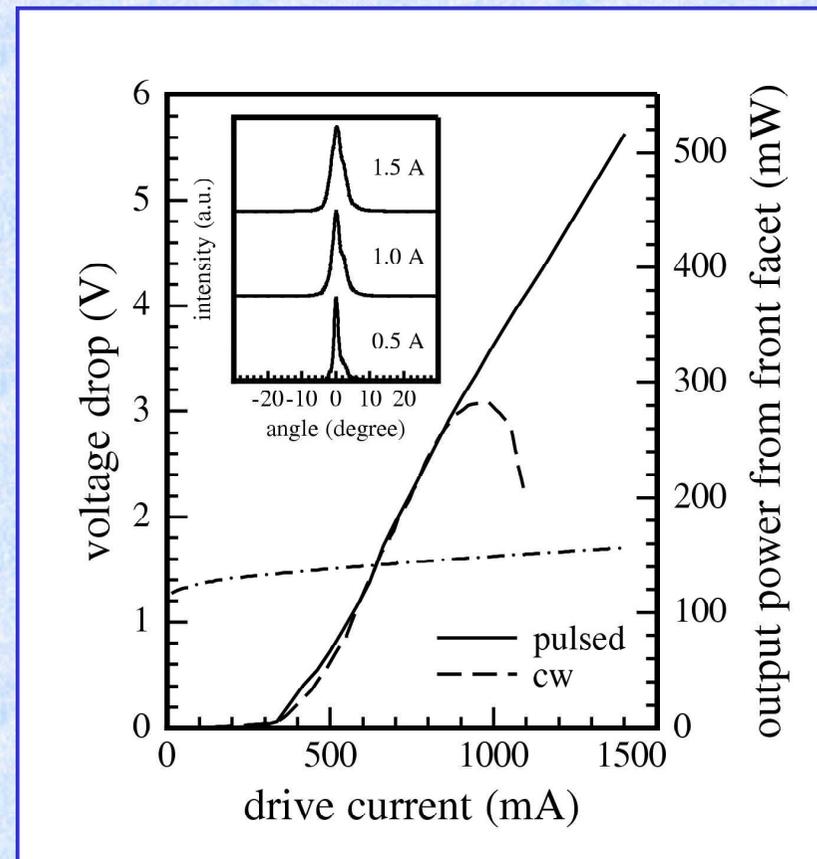
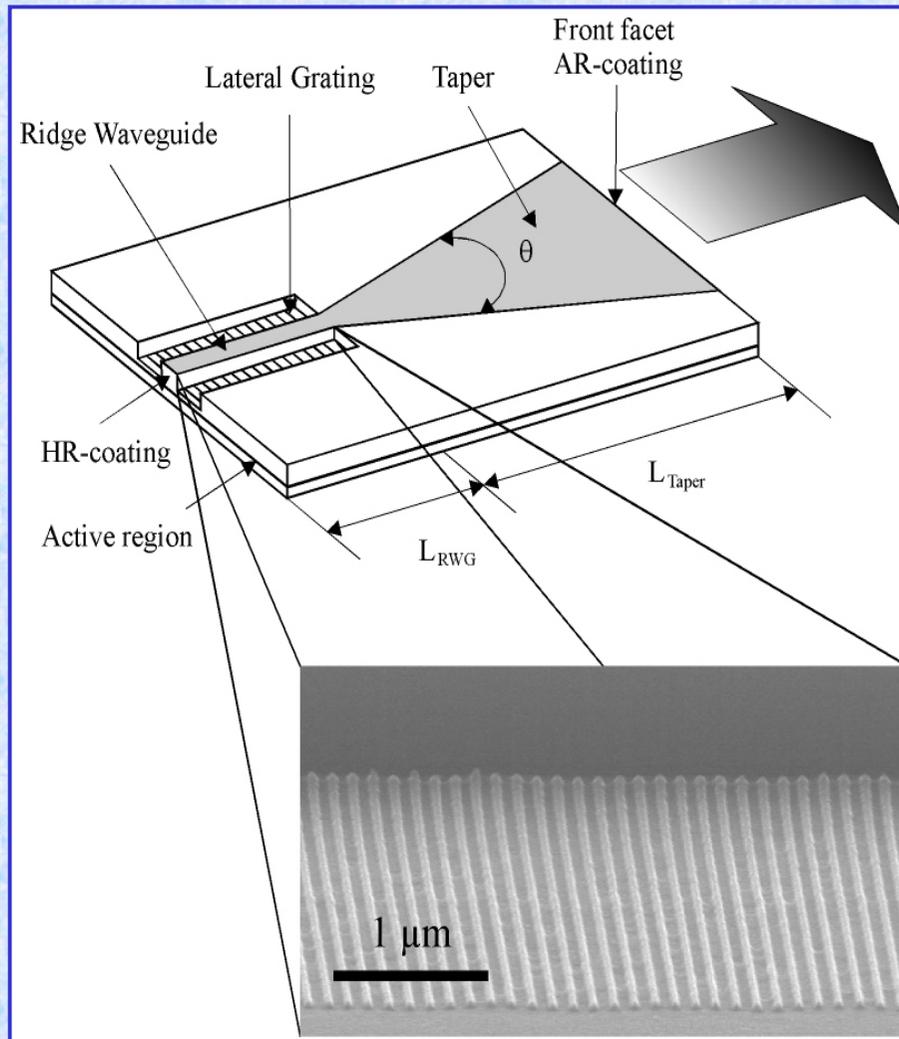


- High single lobe output power
- $M^2 = 2.4$  at 1 W output power
- About 4 times higher brilliance than BA-lasers ( $50 \text{ MWcm}^{-2}\text{sr}^{-1}$ )

- Tapered lasers with  $L_R = 1 \text{ mm}$ ,  $L_T = 2 \text{ mm}$ ,  $\phi = 6^\circ$
- $P_{\max} = 3 \text{ W}$ , wall plug eff. up to 39%

W. Kaiser et al, to be published

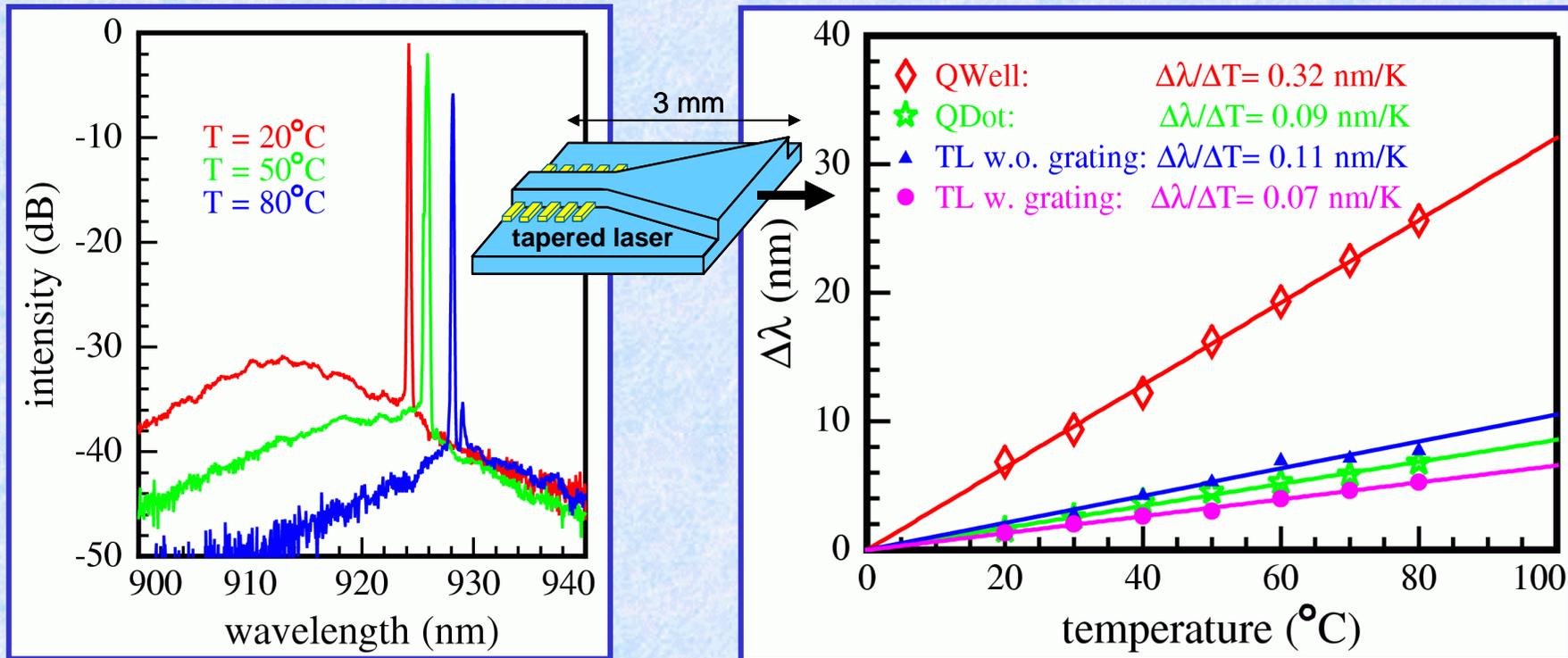
# Single Mode QD Tapered Lasers (920 nm)



- Lateral gratings fix emission wavelength
- Single mode emission up to 500 mW

W. Kaiser et al, to be published

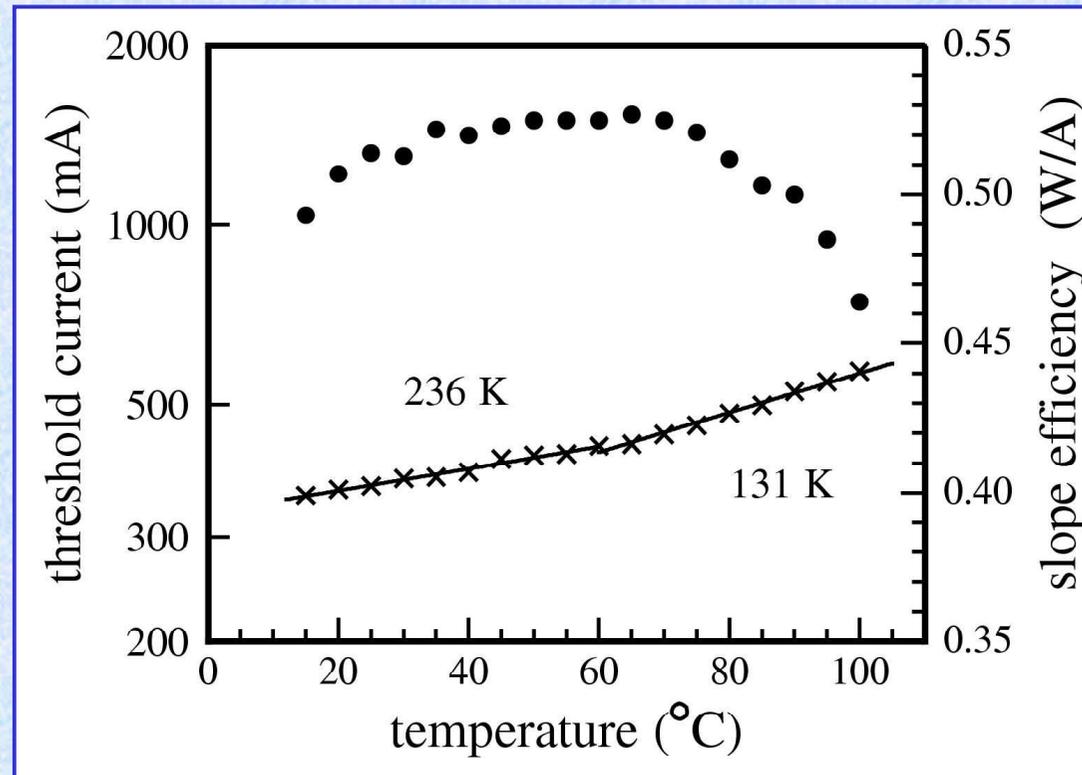
# Wavelength Stabilized QD Lasers with Gratings



- Tapered lasers with lateral gratings
- Single mode emission between 20 to 80 °C due to temperature insensitive gain function
- Temperature stable emission wavelength (0.07 nm/K)
- Temperature dependence of QD material at same order than refractive index change

Kaiser et al., CLEO, Long Beach (May, 2005)

# Temperature Stable Laser Performance (920 nm)

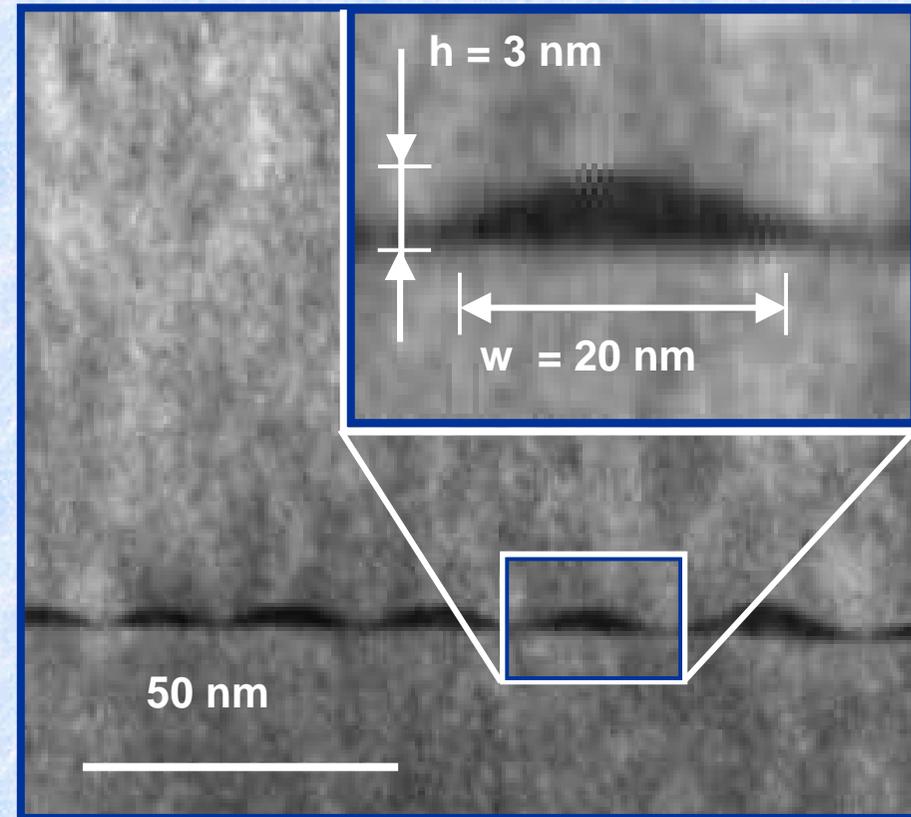
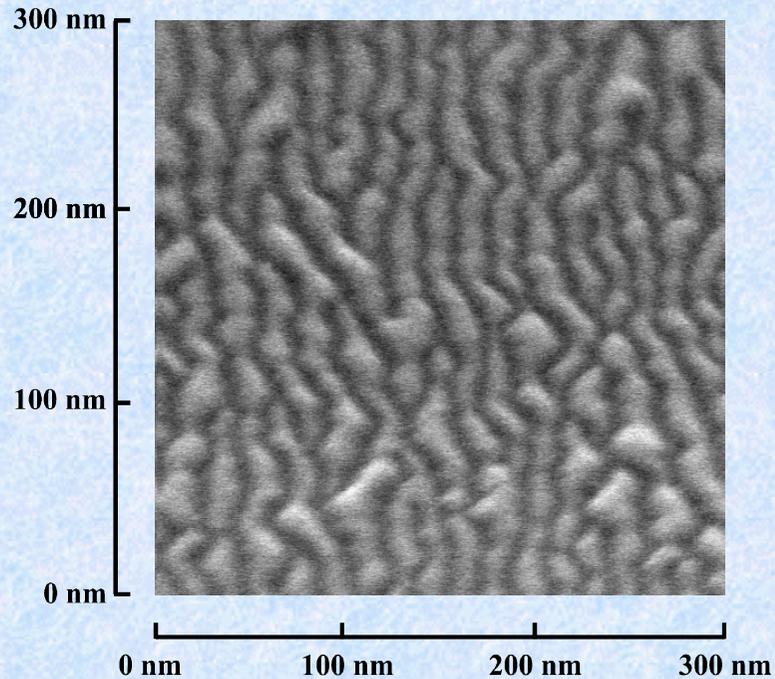


- Constant slope efficiency between 20 – 80 °C
- High  $T_0$  value up to operation temperatures of 100 °C
- Improvement also due to nearly coincident temperature development of gain and grating period

W. Kaiser et al, to be published

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  - ultra-fast broadband SOAs (1.55  $\mu\text{m}$ )

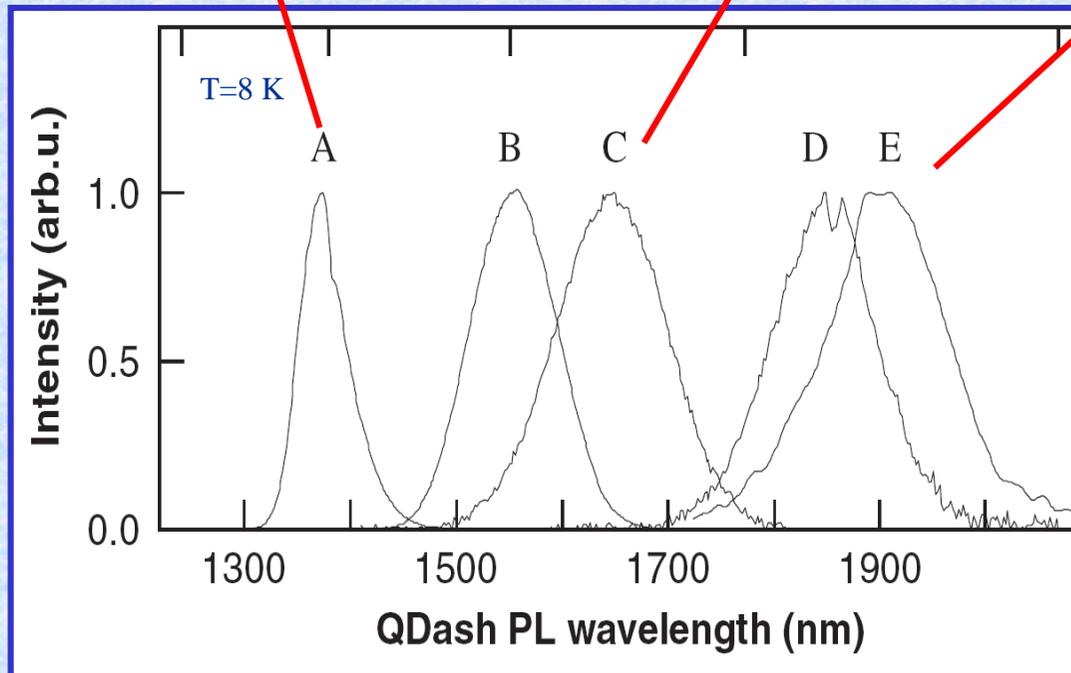
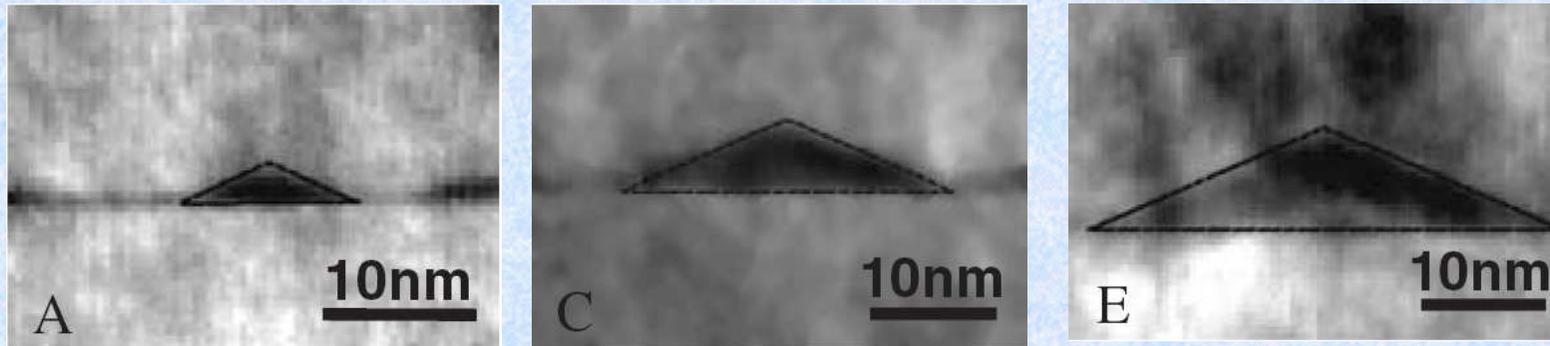
# QDashes on (001) InP-Substrates



- XTEM images done by Univ. of Duisburg (T. Kümmell, G. Bacher)
- Pyramidal like cross-section with dominating vertical quantization

T. Kümmell et al., APL 86, 253112 (2005)

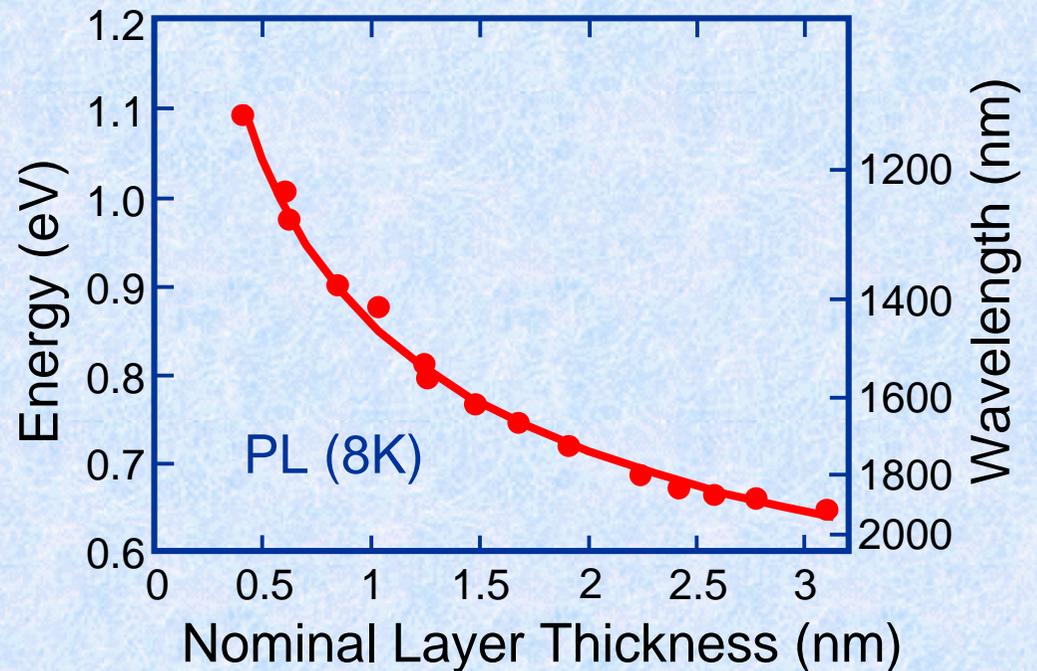
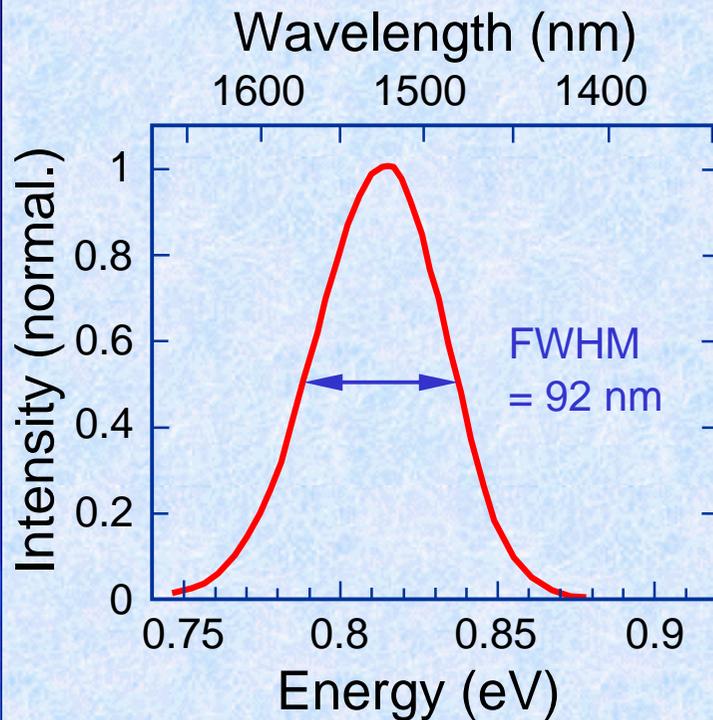
# Shape Preserving QDash Formation Process



- Deposition from 0.8-3.1 MLs (A-E) → wavelength shift > 500 nm
- STEM: Width/height ratio preserved

T. Kümmell et al., Physica E 32, 108 (2006)

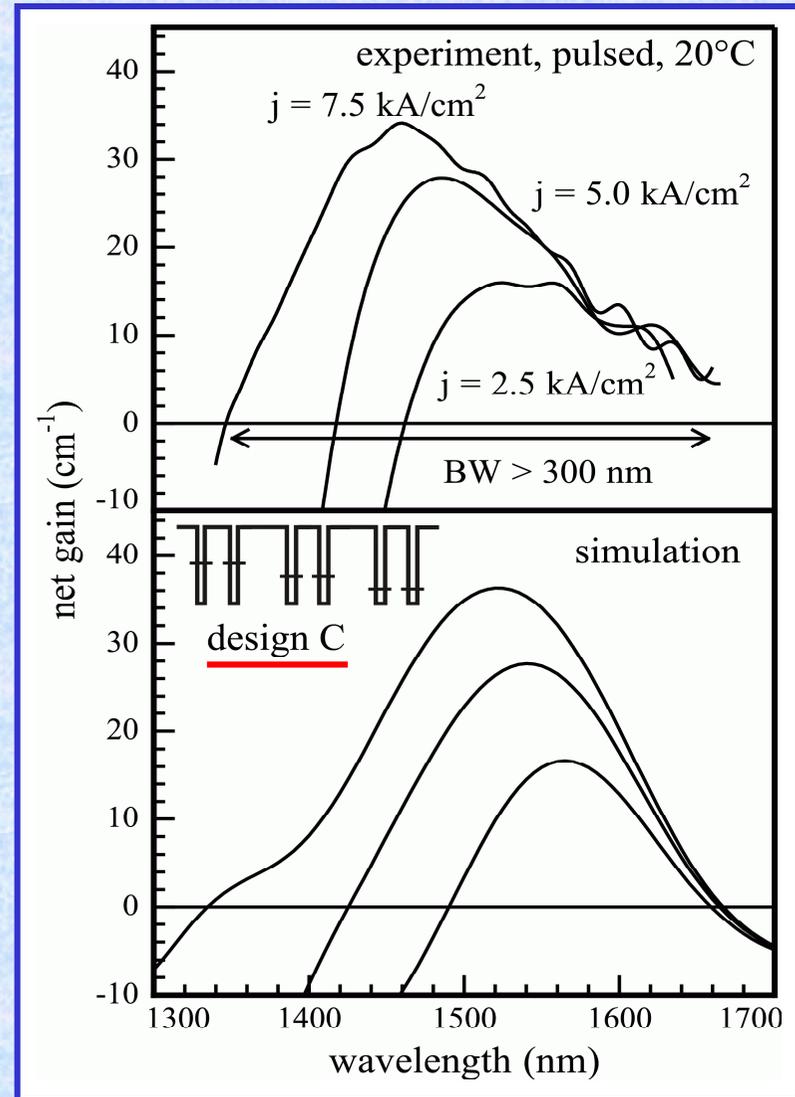
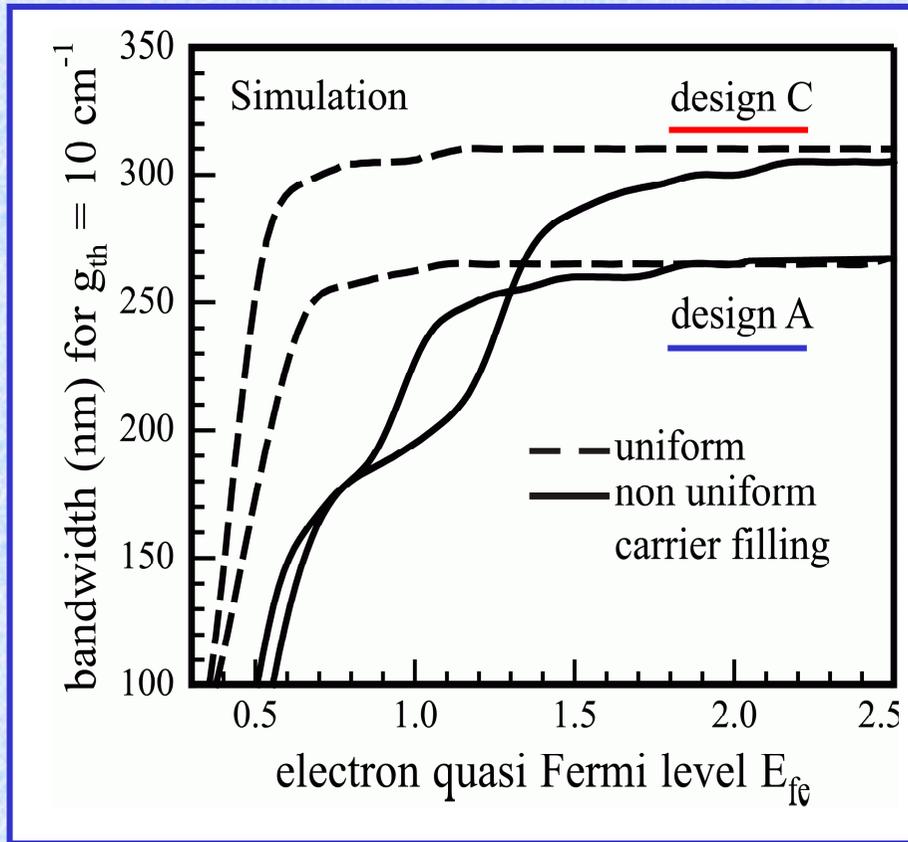
# Variability of QDash Structures on InP



- Nearly symmetric broadband PL at 10 K
- Control of emission wavelength by QDash layer
- Extremely wide wavelength range: 1.2 – 2  $\mu\text{m}$  (at RT)
- By multiple QDash layers ultra-broadband amplification possible

A. Somers, IPRM, Glasgow (2004)

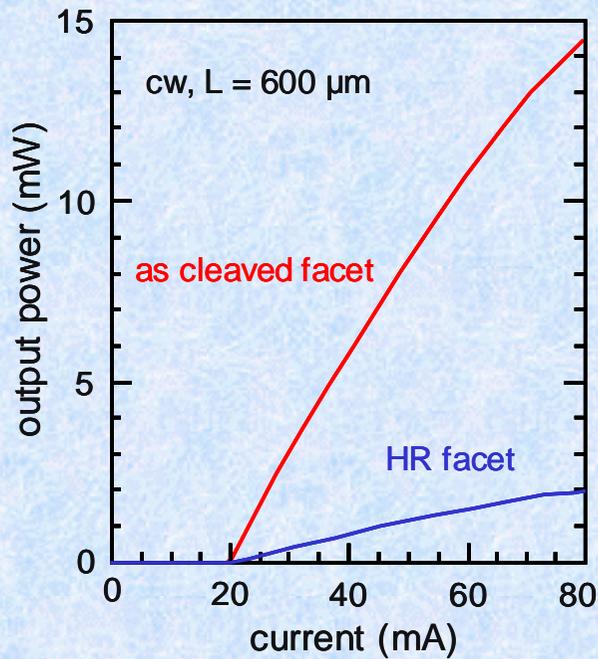
# Ultra-Broad Gain Spectrum



A. Somers et al., APL 89, 061107 (2006)

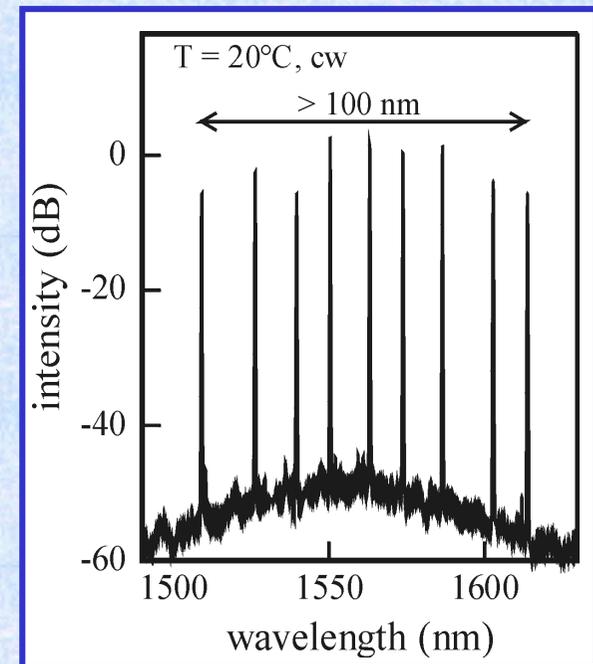
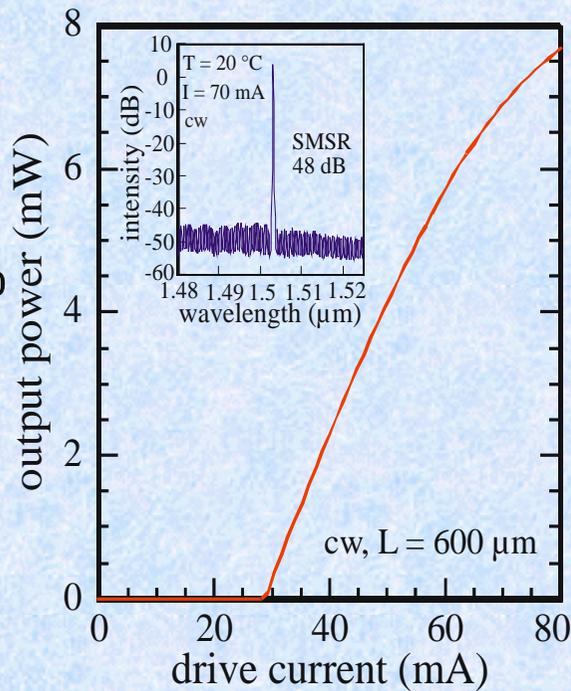
# Static DFB Laser Data

RWG laser with  
HR coated  
backside mirror



Single wafer  
wavelength tuning

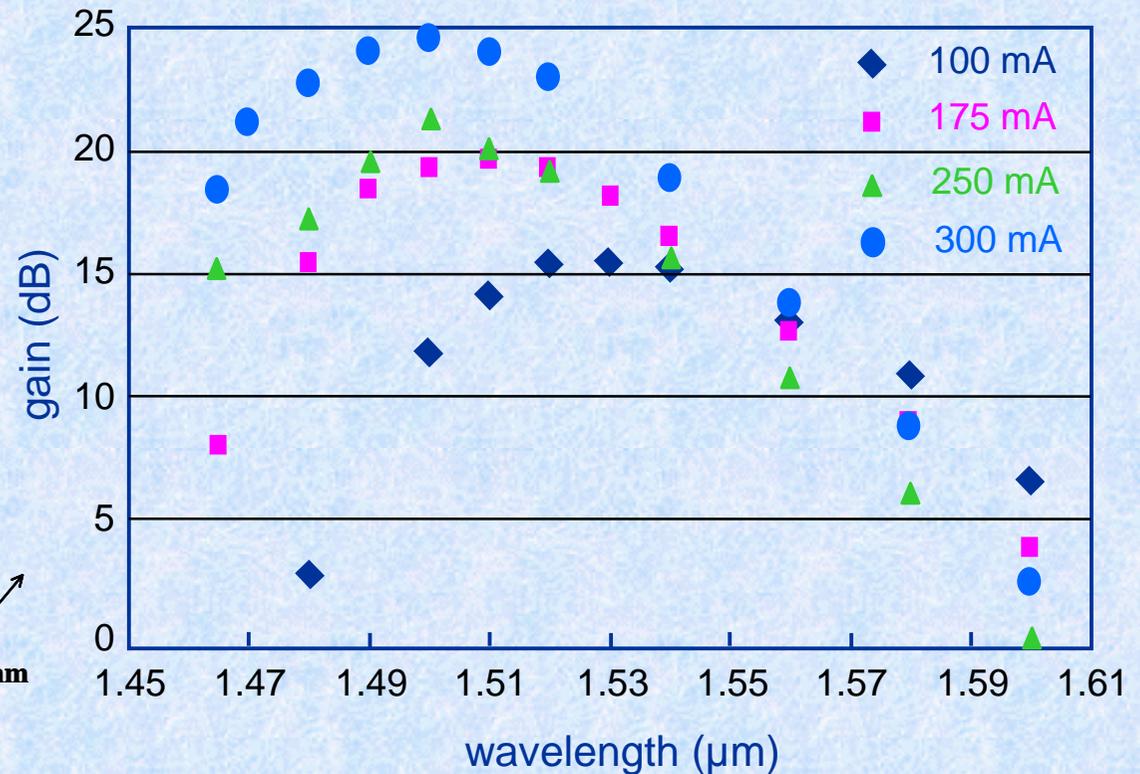
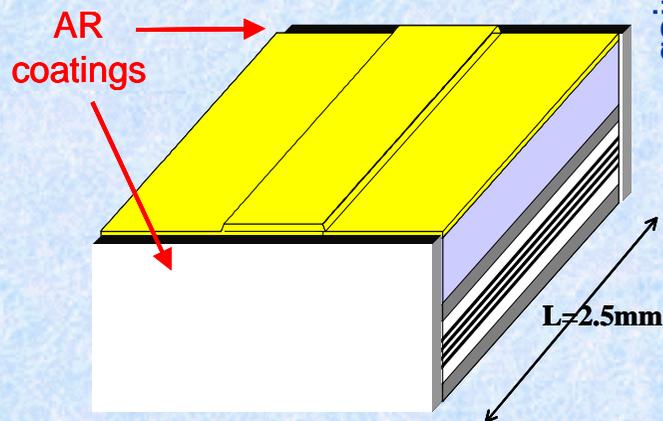
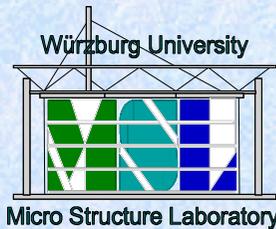
DFB laser with Cr  
gratings



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# Spectral gain of QD-SOAs

THALES

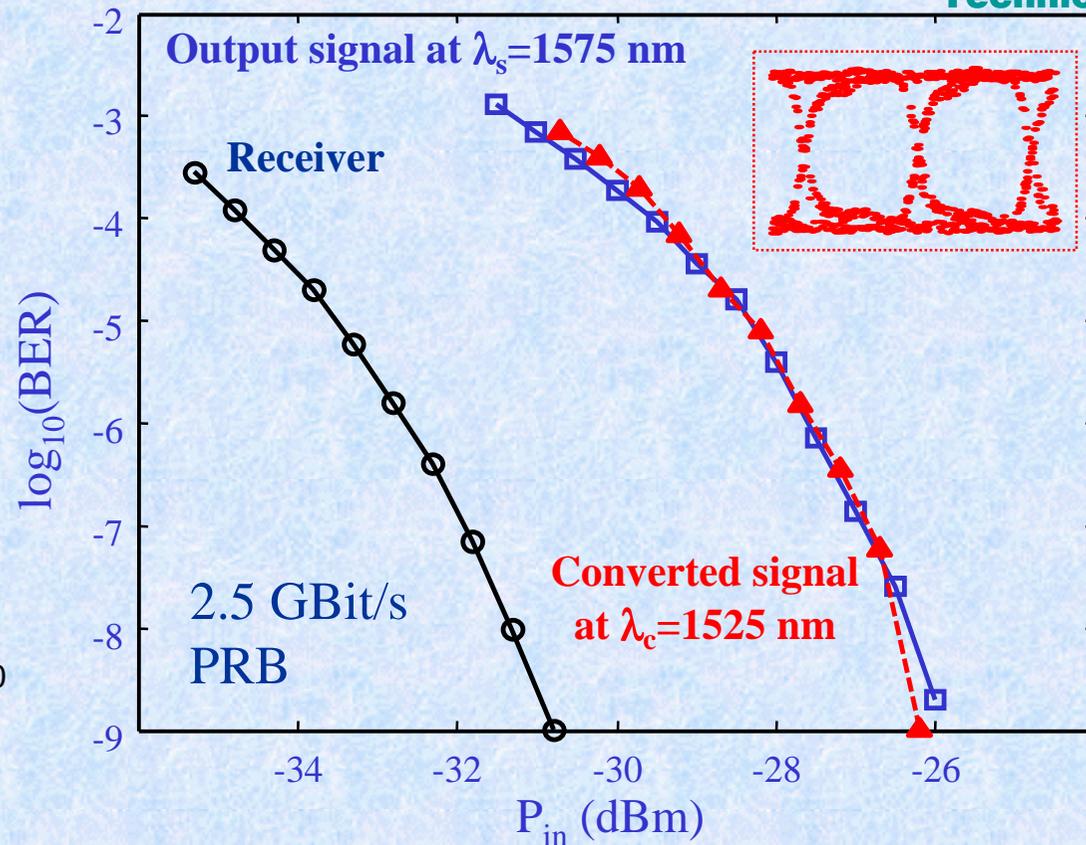
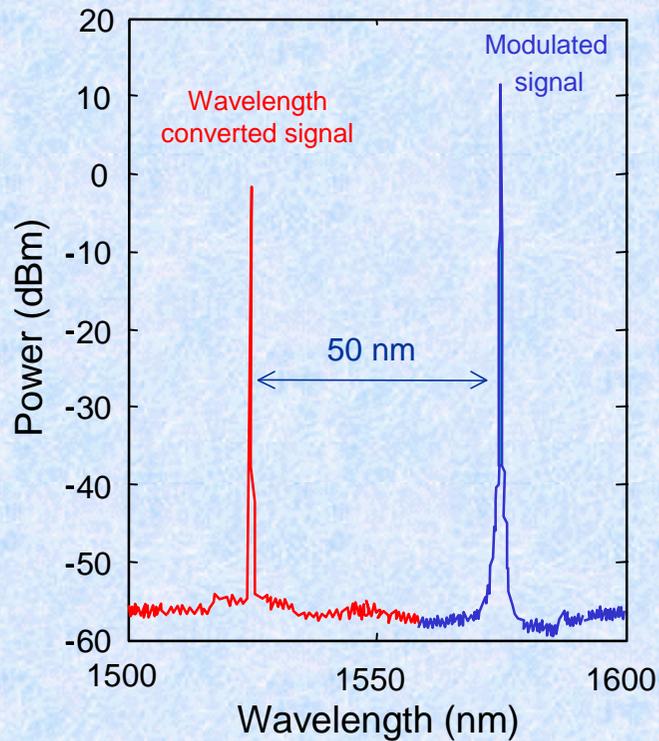


- SOA = RWG laser with anti-reflection coatings
- Large bandwidth ( $> 200$  nm, measurements limited by tunable laser)
- High amplification (25 dB for 2.5 mm long device)
- High saturation power of 18 dBm (= 63 mW)

# BER-Measurements for XG-Modulation



Technion



- About 5 dB penalty due to additional noise of amplifier and set-up
- 50 nm wavelength conversion with 2.5 Gbit/s (open eye diagram)
- BER identical between modulated and **converted** signal

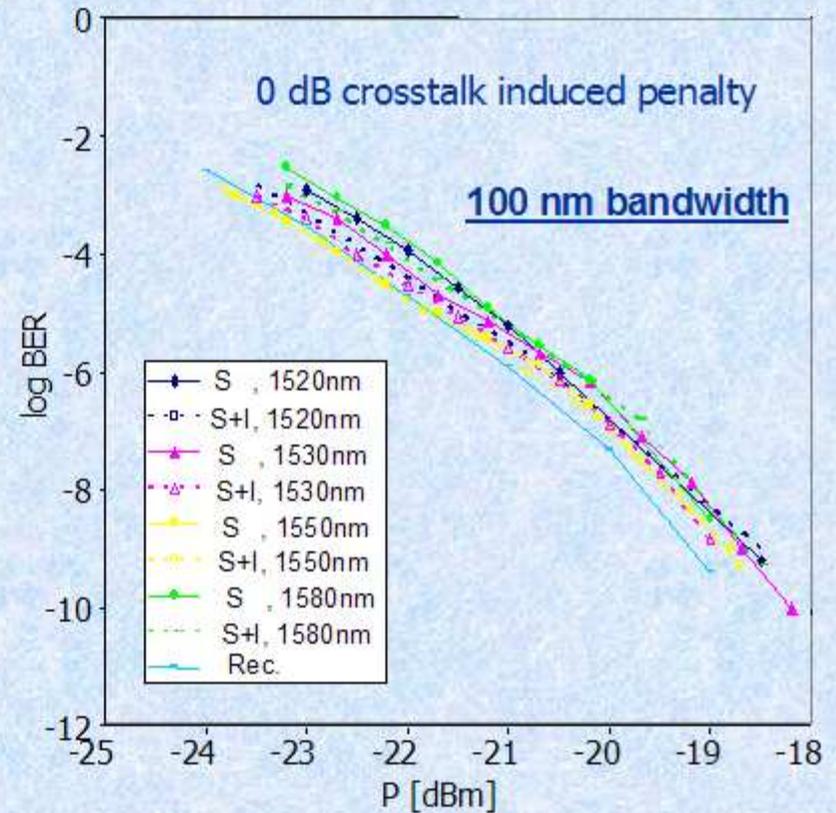
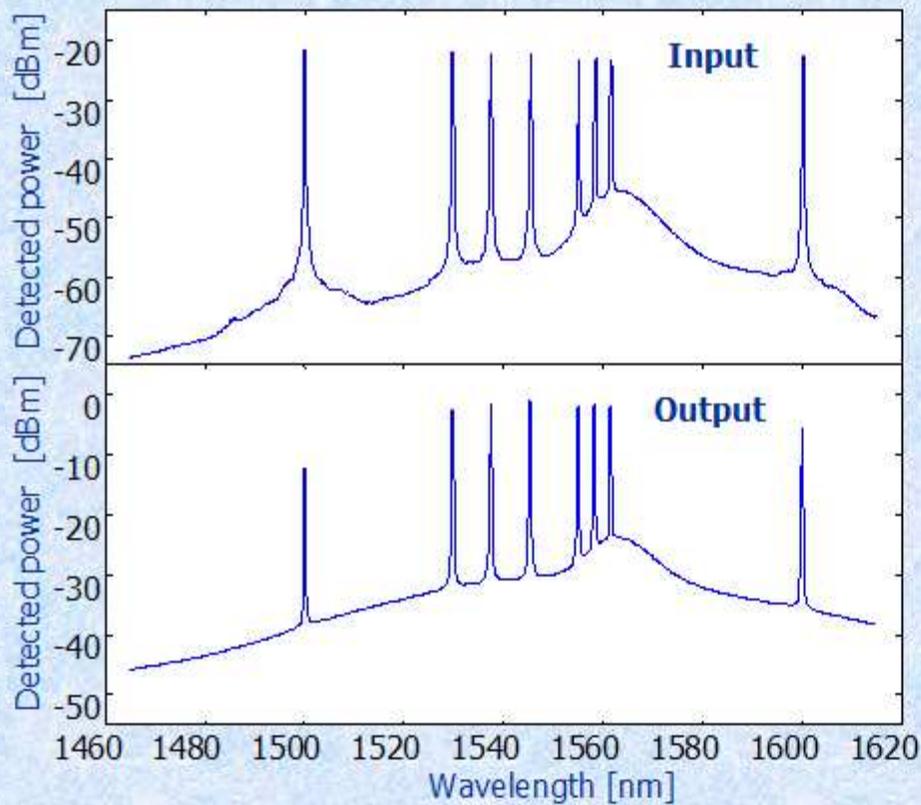
A. Bilenca et al., PTL 15, 563 (2003)

# Multi-wavelength amplification with QD-SOA



Technion

Amplification of eight 10 Gbit/s channels with no cross talk



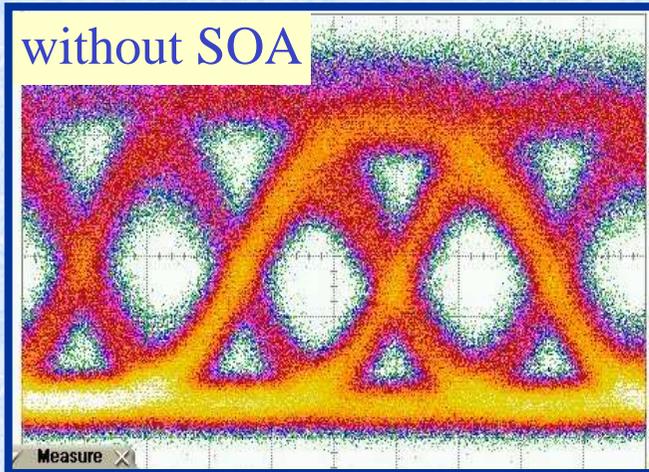
- Amplification of 8 channels over 100 nm at 10 Gbit/s
- $P_{in}$  -21 dBm / channel - 0 dB crosstalk-induced penalty

R. Alizon et al., EL 40, 760 (2004)

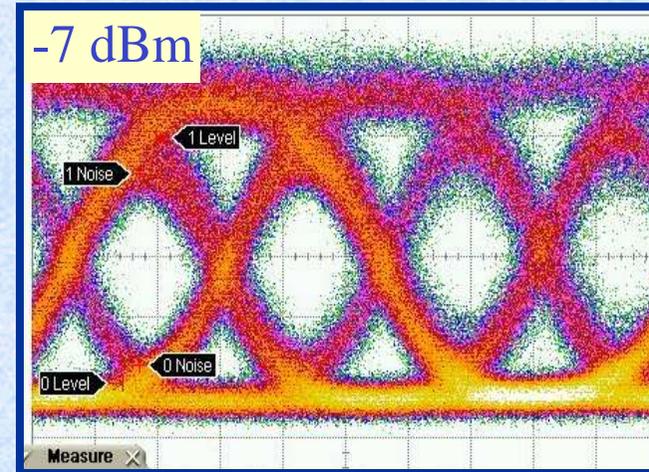
# HF Properties (40 Gbit/s)



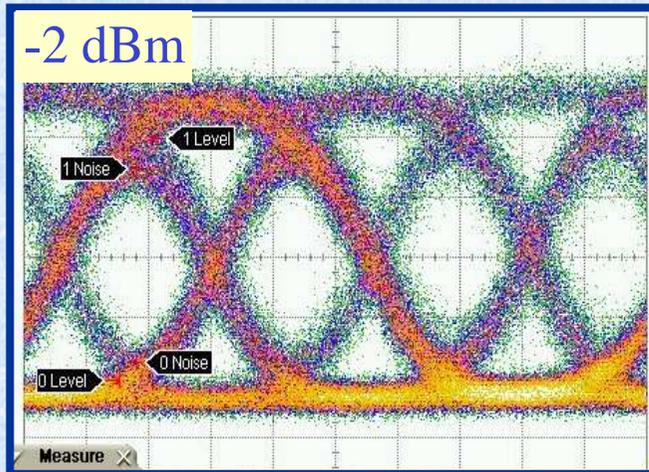
Technion



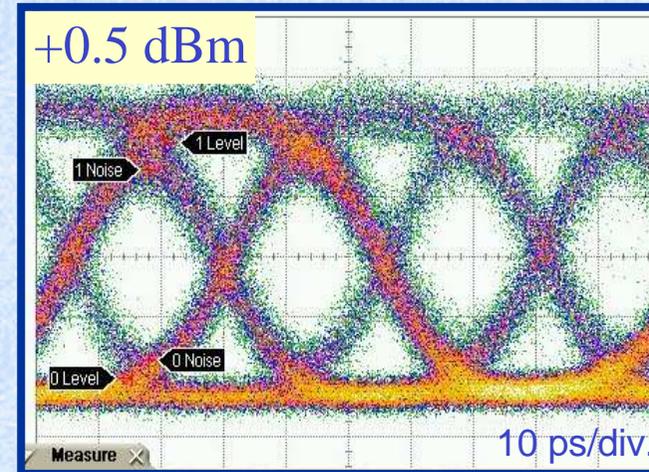
ER 6.3  
Q 4.15



ER 6.2  
Q 4.5



ER 6.1  
Q 5.0



ER 5.9  
Q 5.3

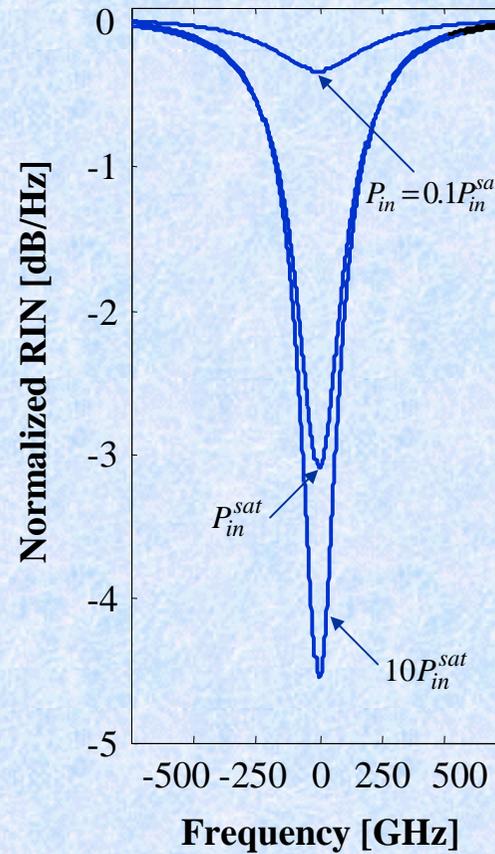
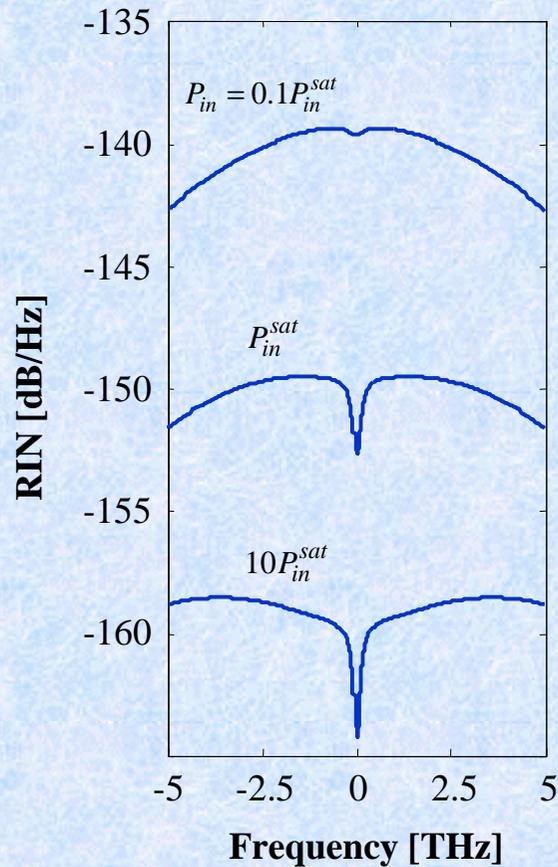
- 40 Gbit/s PRB input signal: no patterning effect
- Improved signal for operation in saturation condition

J.P. Reithmaier et al.,  
JPD 38, 2088 (2005)

# Calculated RIN Spectra



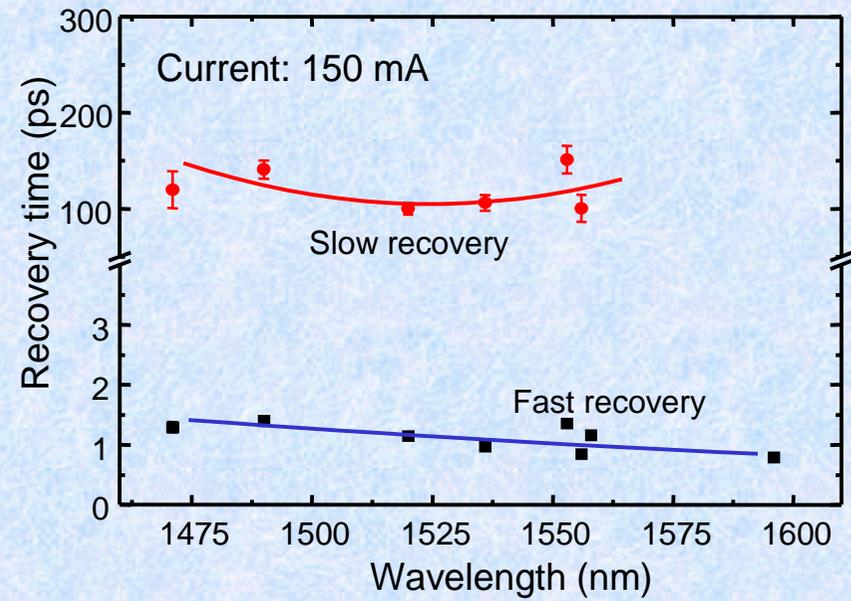
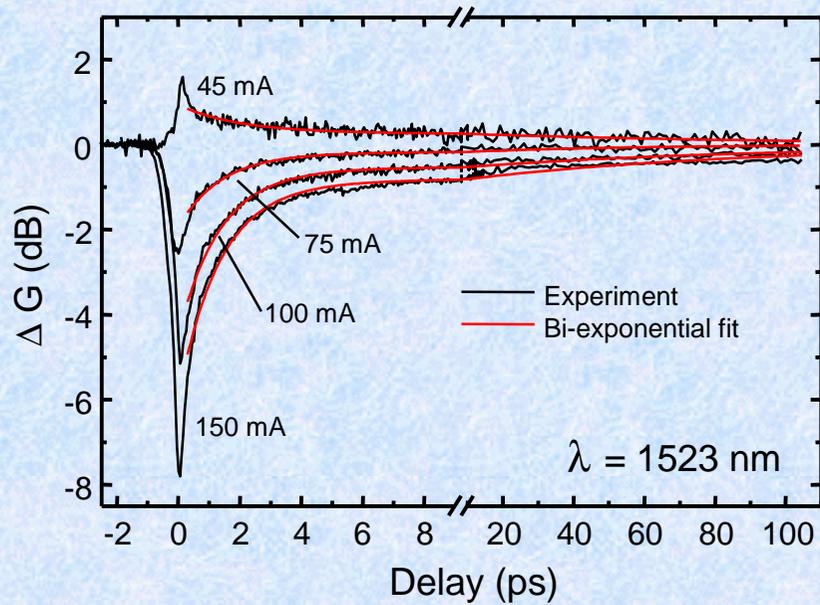
Technion



- Reduction in RIN signal due to band filling
- Broad band intensity noise suppression over hundreds of GHz !

D. Hadass, JSTQE 11, 1015 (2005)

# Pump-Probe Measurements

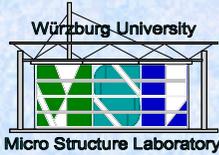


$\tau_1 = 1-2$  ps (local carrier storage)  
 $\tau_3 > 100$  ps (transport time)

# Summary

- **Theoretical Background of Low Dimensional Systems**
  - Strong influence of dimensionality on gain properties
  - Additional geometry parameters can be used for spectral gain engineering and tailoring of new material properties
- **Fabrication Technology of Dot-Like Structures**
  - Self-assembly techniques driven by material strain
  - Geometry parameters (density, size, size distribution) can be controlled by growth parameters
- **High power QD lasers**
  - BA laser: 6.3 W (980 nm), 3 W (920 nm) output power,  $\eta_w = 55\%$
  - Tapered laser: 3 W cw single lobe output power,  $\eta_w = 39\%$
  - Internal temperature compensation by dot tailoring  
(BA laser:  $d\lambda/dT = 0.11$  nm/K, T laser:  $d\lambda/dT = 0.07$  nm/K)
- **1.55  $\mu\text{m}$  QD Lasers and SOAs**
  - ultra-broad band gain material ( $> 300$  nm) for telecom laser appl.
  - 10 GBit/s multi-wavelength amplification
  - 40 GBit/s pattern free signal amplification and recovery

# Acknowledgements



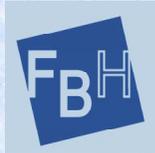
University of Würzburg



Thales R & T

III-V lab

Alcatel – Thales III-V Lab



Ferdinand Braun Institut  
für Höchstfrequenztechnik



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## European IST-Projects



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